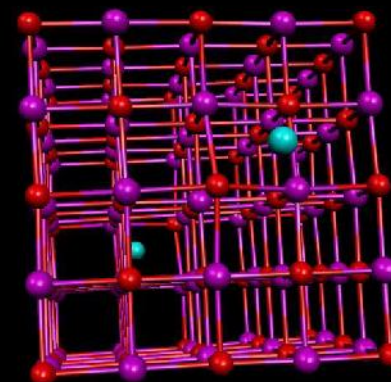


Multiscale modeling in design of novel engineering materials

Modelowanie wieloskalowe w projektowaniu nowoczesnych materiałów inżynierskich

Dr. Tomasz Wejrzanowski
Warsaw University of Technology
Faculty of Materials Science and Engineering

Dr inż. Romuald Dobosz
Dr inż. Maciej Spychalski
Dr inż. Marek Muzyk
Mgr inż. Jakub Skibiński
Mgr inż. Mateusz Grybczuk
Jakub Szabelewski



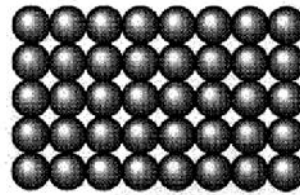
Outline

1. Introduction
2. Methods
3. Design of materials properties
4. Design of materials fabrication processes
5. Summary

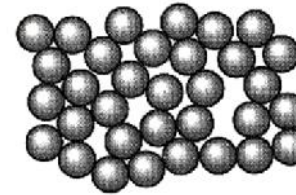


Complexity of materials structure

atomic structure



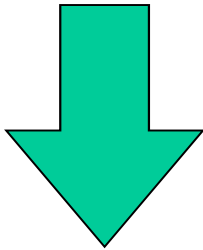
Crystalline solid



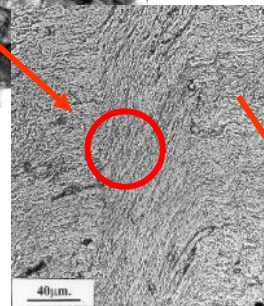
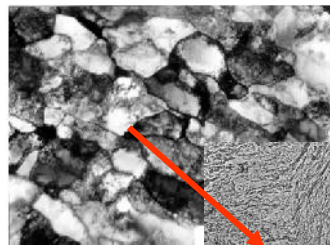
Amorphous solid

Features:

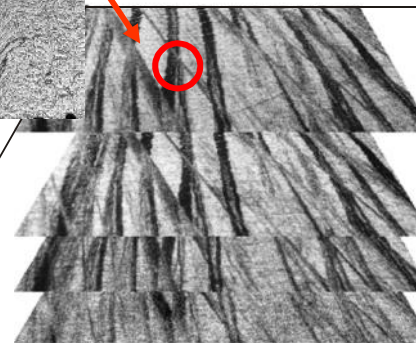
phase composition,
bonding energy,
lattice size, density,
electron density



(nano, microstructure)



1 mm



structural elements:
volume fraction,
surface and number
per unit volume, size
, shape, orientation





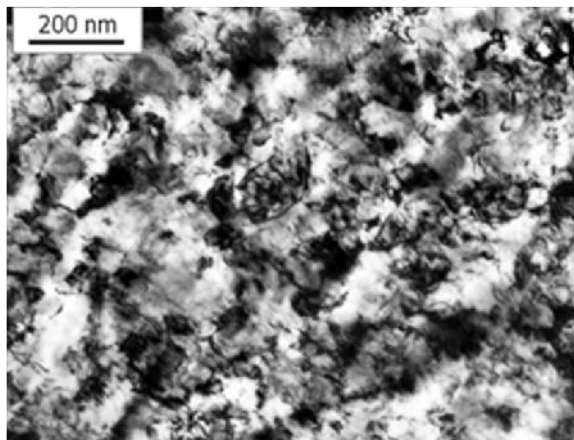
Complexity of materials structure



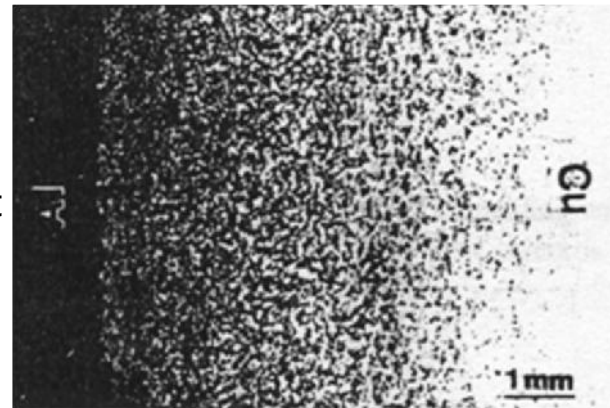
microstructure



anisotropic:

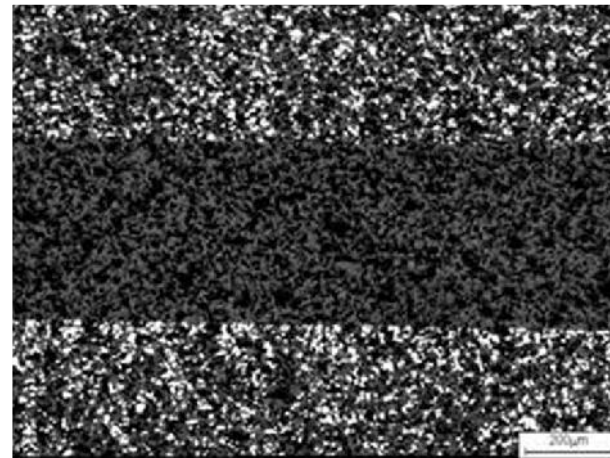


- gradient (FGM)



spacial
arrangement of
structural
elements

- layers (films)



isotropic

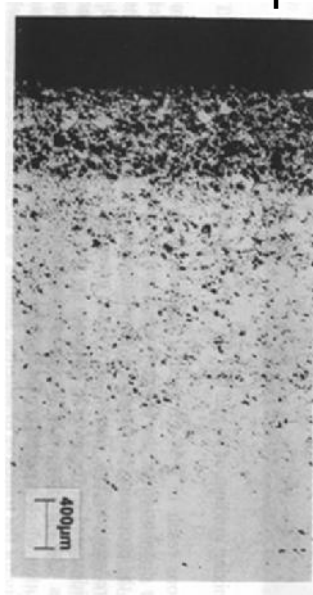




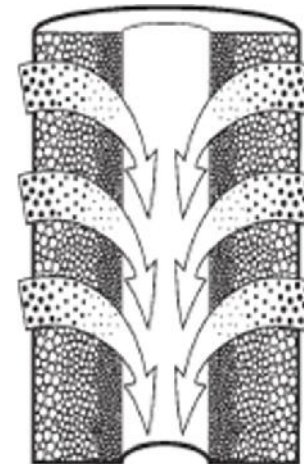
Complexity of materials structure

microstructure

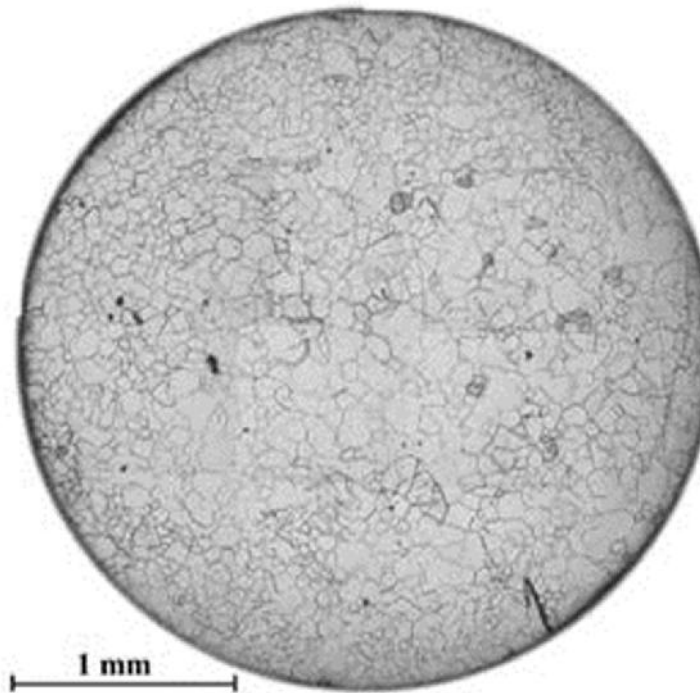
Composition gradient



Porosity gradient



*Graded Density
Polymicro Filters trap
contaminants
throughout the entire
cross section for
improved efficiency.*

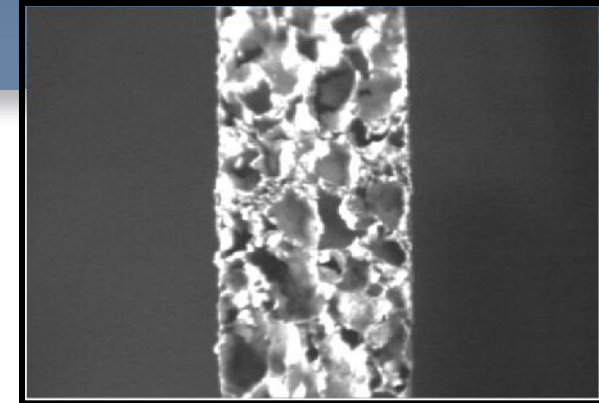


Grain size gradient

Complexity of materials structure

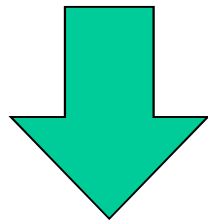
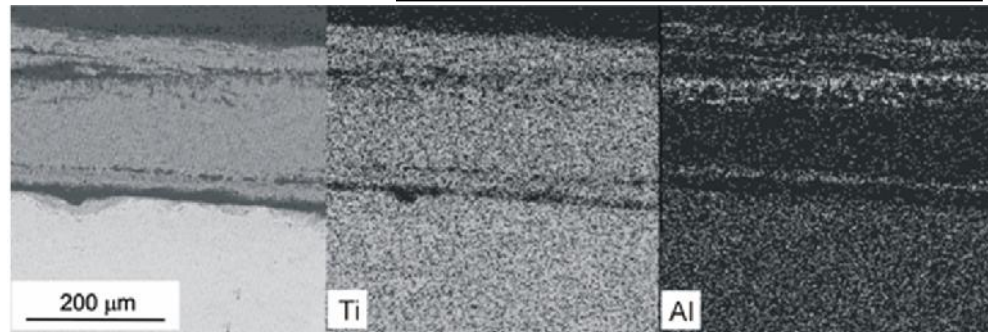


Bulk

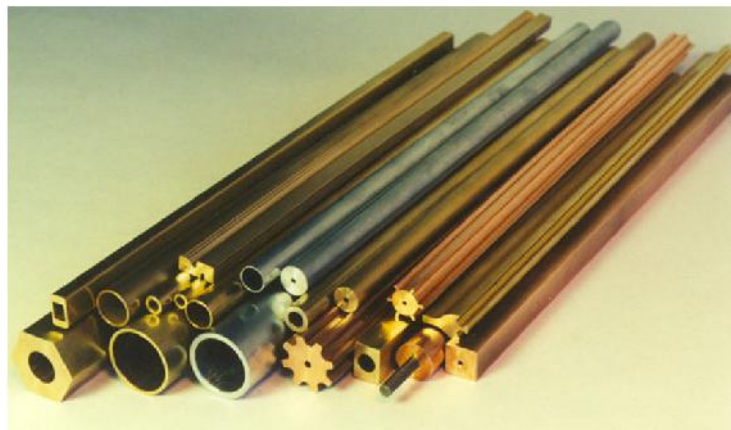


macrostructure

Films



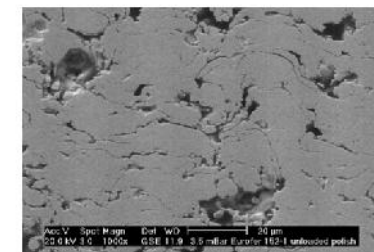
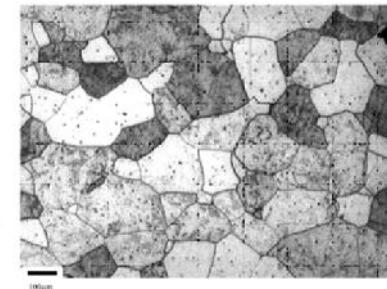
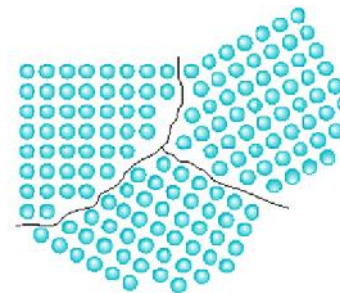
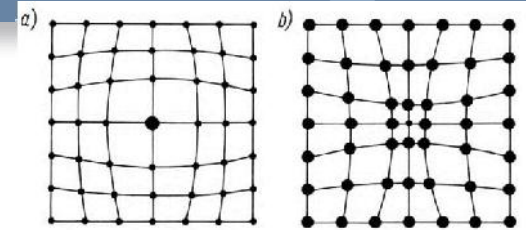
**engineering
materials**



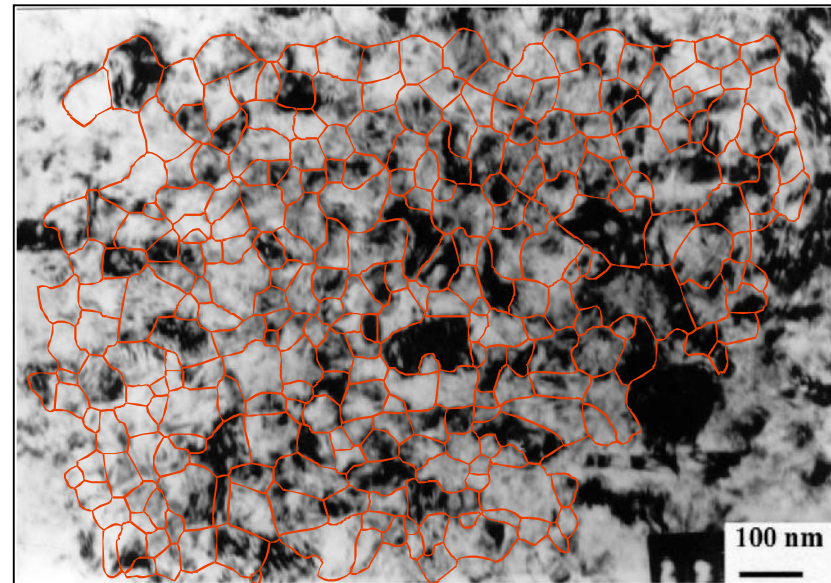
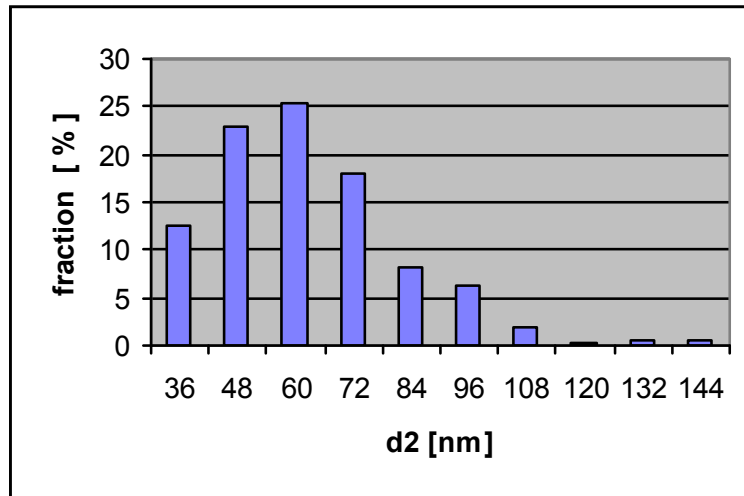
Defects in materials

Dimensionality based classification:

- 0D – Point defects
 - vacancy
 - Foreign atom
- 1D – Linear defects
 - dislocation
 - triple junction (triple point)
- 2D – Surface defects
 - grain boundary
 - phase boundary
- 3D – Volume defects
 - pore
 - particle
 - inclusion



- Population of defects described by stochastic variables



mean diameter = 55 nm
 standard deviation of diameter = 37 nm
 coefficient of deviation = 0,67

.....



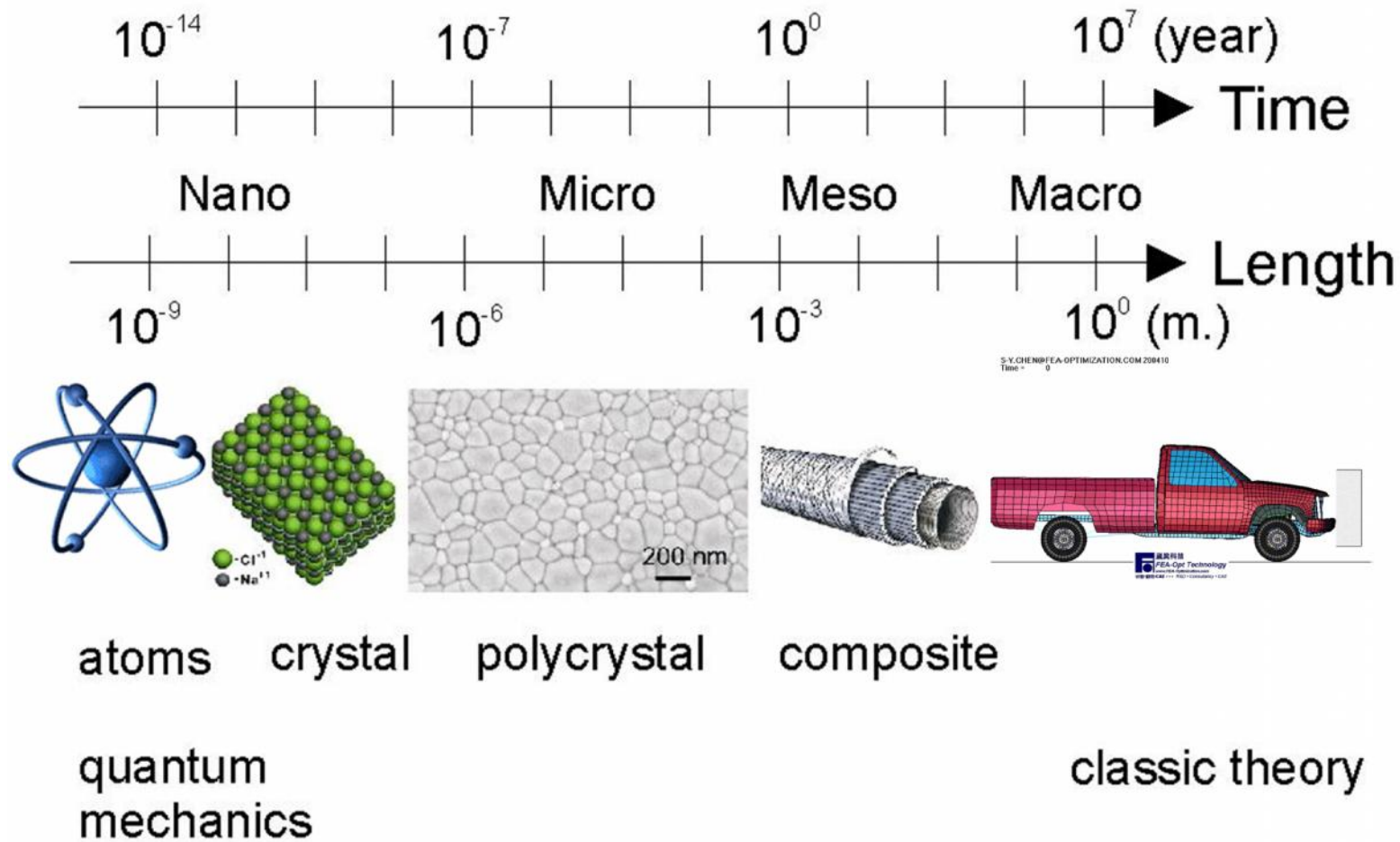
Properties of materials

Properties dependent on atomic structure	Properties dependent on microstructure
electrochemical potential	Corrosion resistance
Melting point	
Specific heat	
Thermoelectrical potential	Resistivity of semiconductors
magnetization	coersivity
Young modulus Density	Strength Ductility
colour	transparency

Quantitative structure description

○ V_V	Hardness
○ S_V	Reactivity
○ d_3	Flow stress
○ $CV(d_3)$	Fracture strength
○ shape factors	Anisotropy
○ Voronoi tessellation	Conductivity

Time and length scale



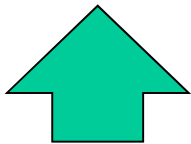
Methods

Mutiscale modelling

Continuum level



Mesoscale



Atomistic scale

FEM

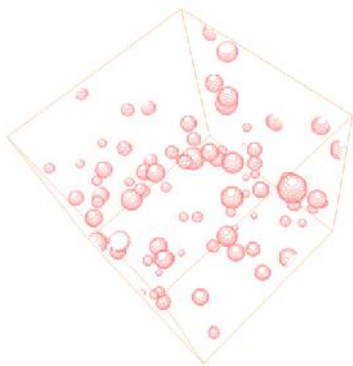
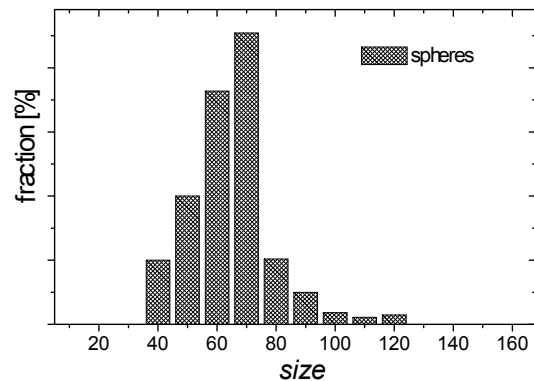
Cellular Automaton
Mesoscale MonteCarlo

Molecular Dynamics
Atomic-scale MonteCarlo
Ab initio

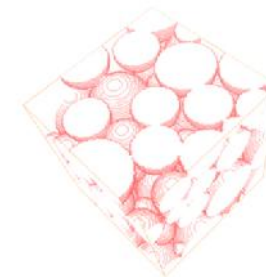
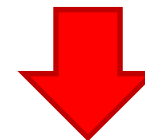
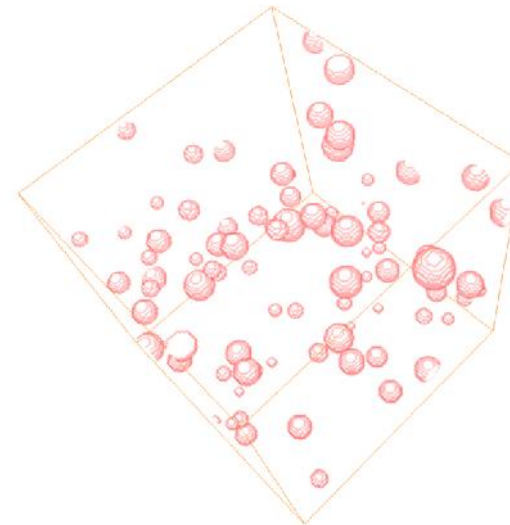
- Macroscopic Strain – Stress relations
- Grain size, shape efect
- Strain localization
- Internal stresses
- Rotation of grains
- Grains sliding
- Dislocation creation and motion
- Elastic constants
- Point defects

Design of materials structures at various scales

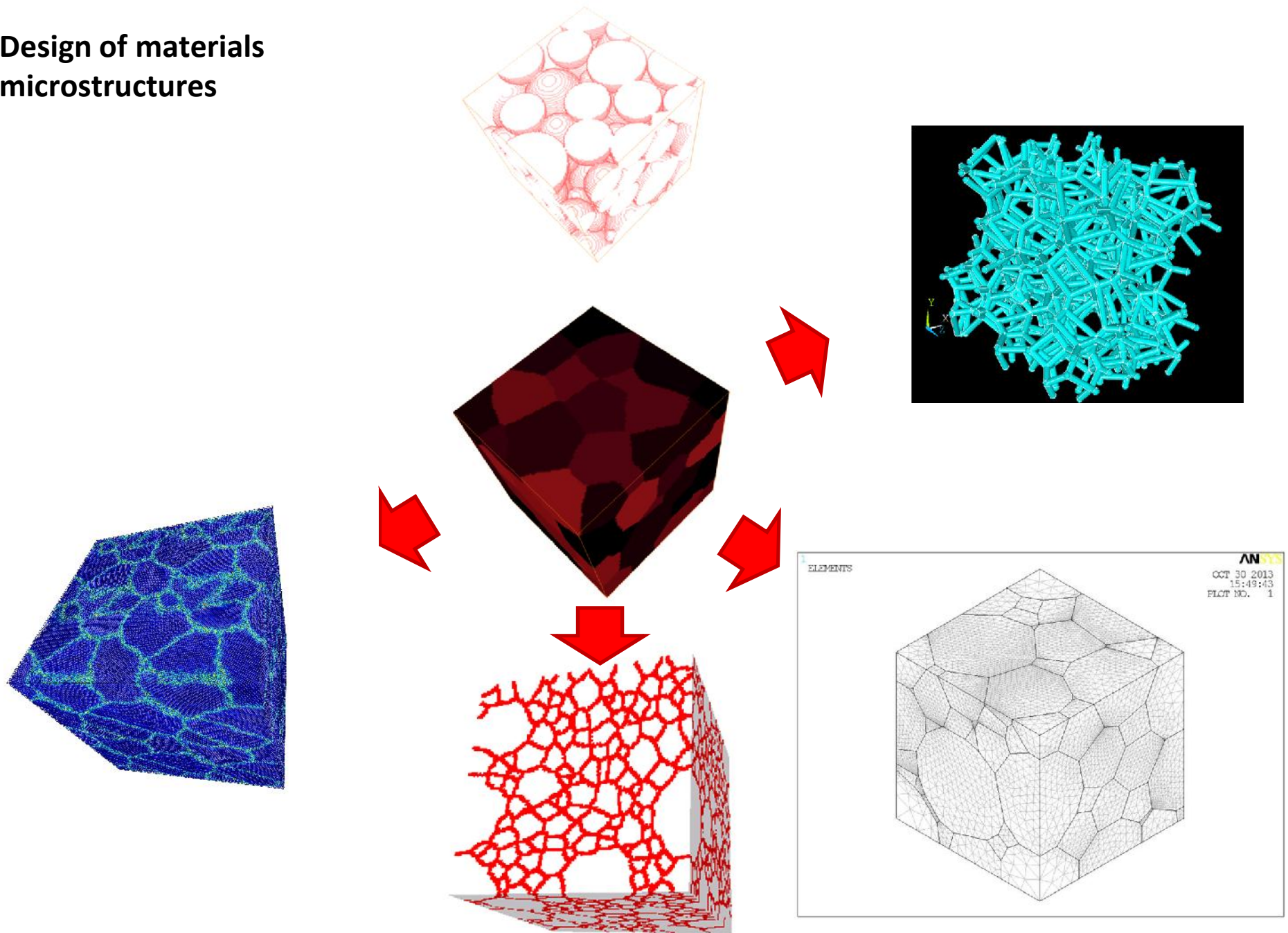
Step 1. Definition of the size distribution of spheres



Step 2. Packing of spheres



Design of materials microstructures



Numerical design of the properties of materials

Nanometals

- Mechanical properties of nanometals – multiscale approach
- Thermal stability – multiscale approach
- Melting point
- Corrosion resistance

Composites

- Thermal conductivity of Cu,Ag – graphene composites

Foams

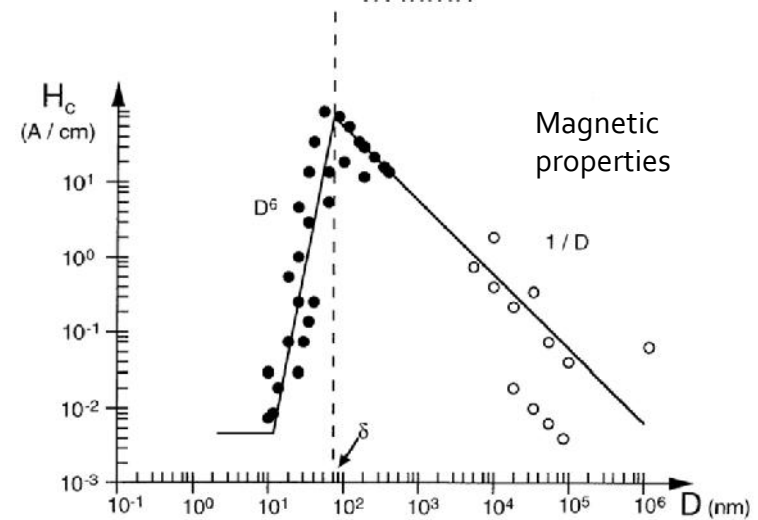
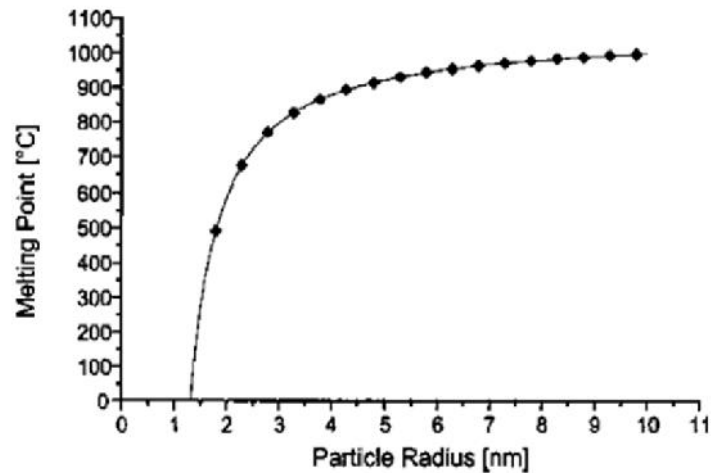
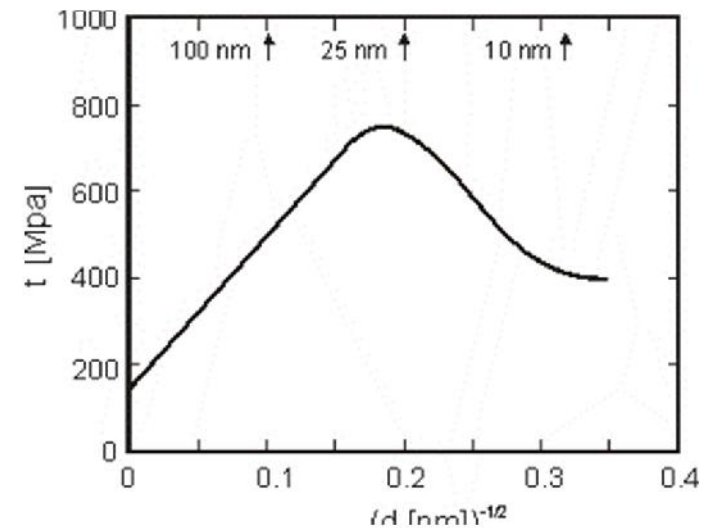
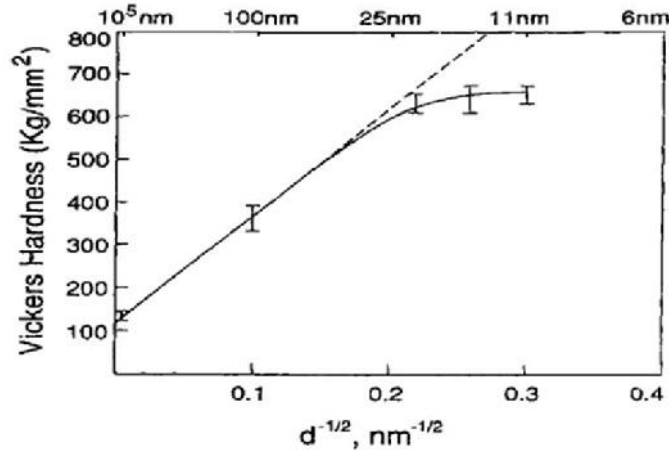
- Permability
- Mechanical properties
- Heat dissipation

Numerical design of the materials processes

- Multipass hot-rolling of steel plates
- GaN and Graphene epitaxial growth
- SiC PVT growth

Application to nanomaterials

Special properties of nanomaterials



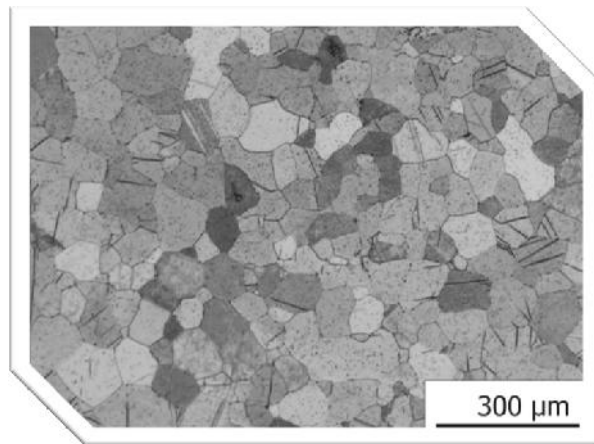
S.C. Tjong, H. Chen / Materials Science and Engineering R 45 (2004) 1–88

Acta Materialia 54 (2006) 297–303

Grain boundaries in polycrystalline materials

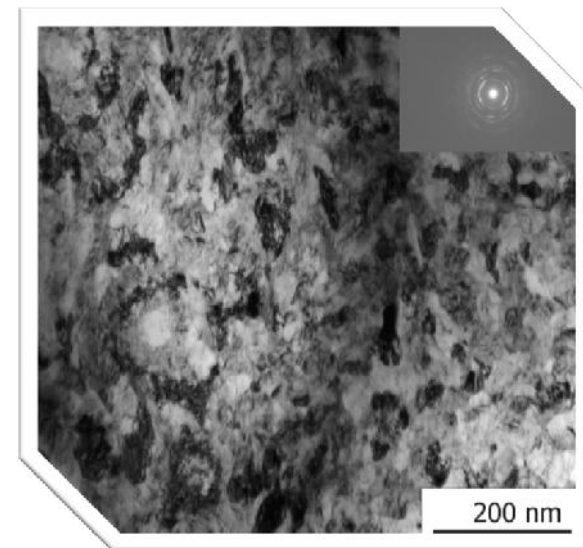
Specific surface of grain boundaries

Microcrystalline



$$S_V = 10^4 - 10^5 \text{ m}^2/\text{m}^3$$

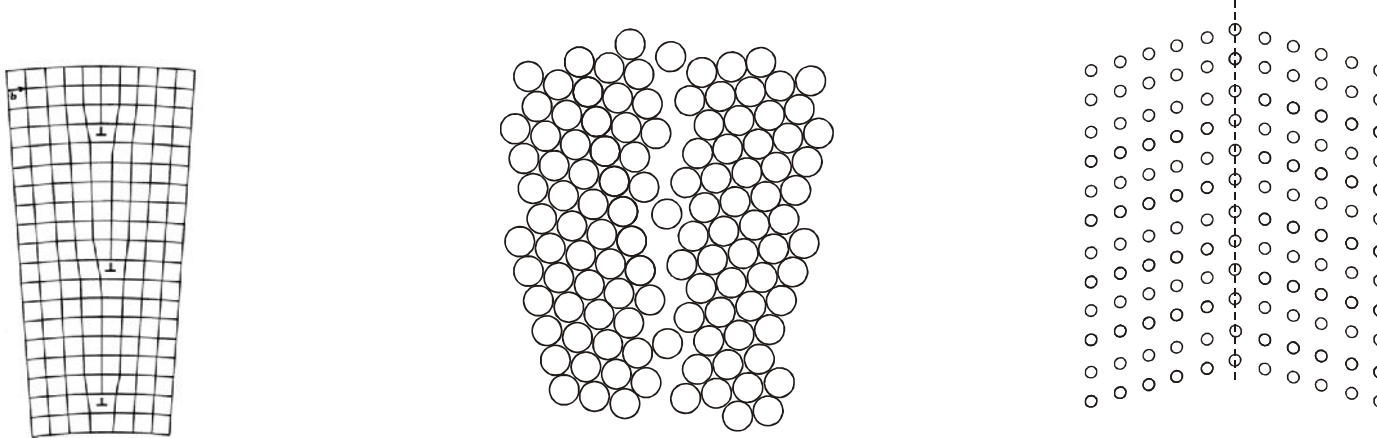
Nanocrystalline



$$S_V = 10^7 - 10^8 \text{ m}^2/\text{m}^3 = 0.1 \text{ m}^2/\text{mm}^3$$

Grain boundaries in polycrystalline materials

Grain boundaries – models



Grain boundaries – types

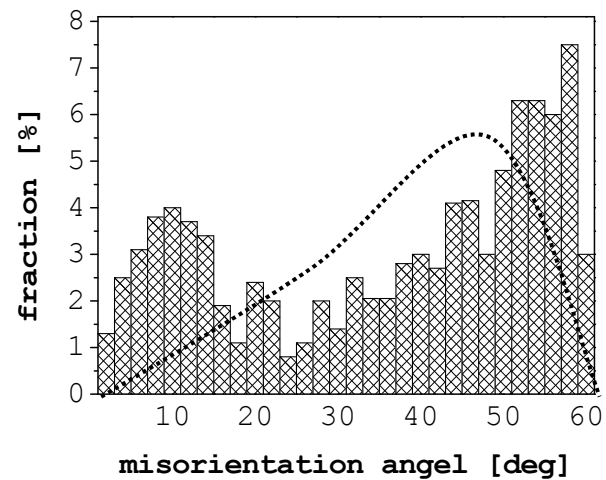
Type of grain boundaries: a) low-angle, b) high-angle, c) twin

1. Random GB
2. Special GB

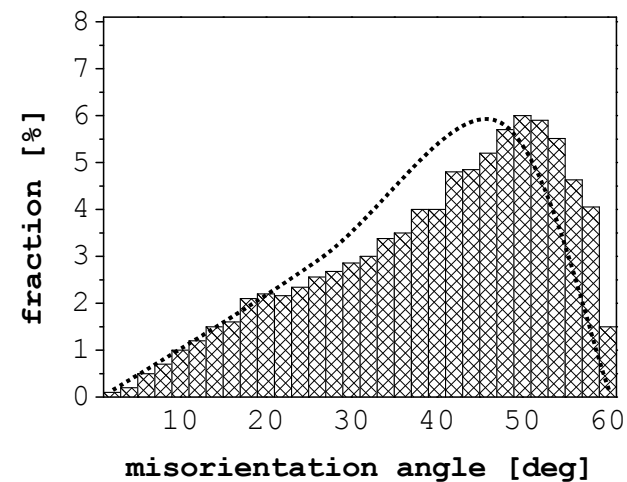
Grain boundaries in polycrystalline materials

Structure of grain boundaries

Grain boundaries – distribution of GB misorientation

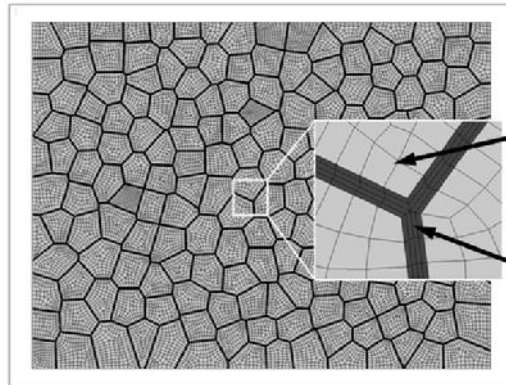


Cu after cold rolling



Cu after cold rolling and long annealing

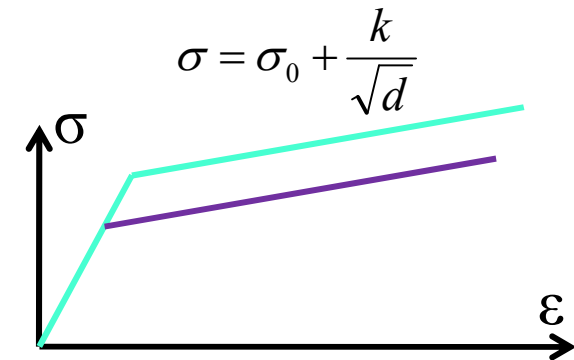
Mechanical properties of nanometals



Data available

Grain Interior:
Plastic Bi-linear Isotropic model
+ Hall-Petch equation for yield
point of each grain

Grain Boundaries:
Anisotropic Hill Potential model



$$[M] = \begin{bmatrix} M_{11} & M_{12} & M_{13} & 0 & 0 & 0 \\ M_{12} & M_{22} & M_{23} & 0 & 0 & 0 \\ M_{13} & M_{23} & M_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & M_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & M_{55} & 0 \\ 0 & 0 & 0 & 0 & 0 & M_{66} \end{bmatrix}$$

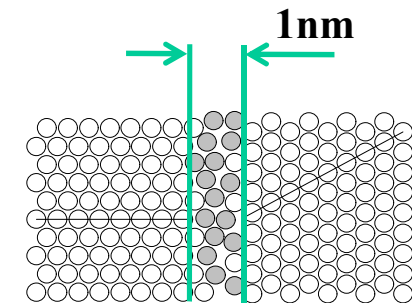
1. GB Young modulus
2. GB shear modulus
3. GB shear stress

Data is missing !!!

$$\{\sigma\}^T [M] \{\sigma\} - \{\sigma\}^T \{L\} - K = 0$$

$$M_{jj} = \frac{K}{\sigma_{+j} \cdot \sigma_{-j}}, \quad j = 1..6$$

$$K = \sigma_{+K} \cdot \sigma_{-K}$$

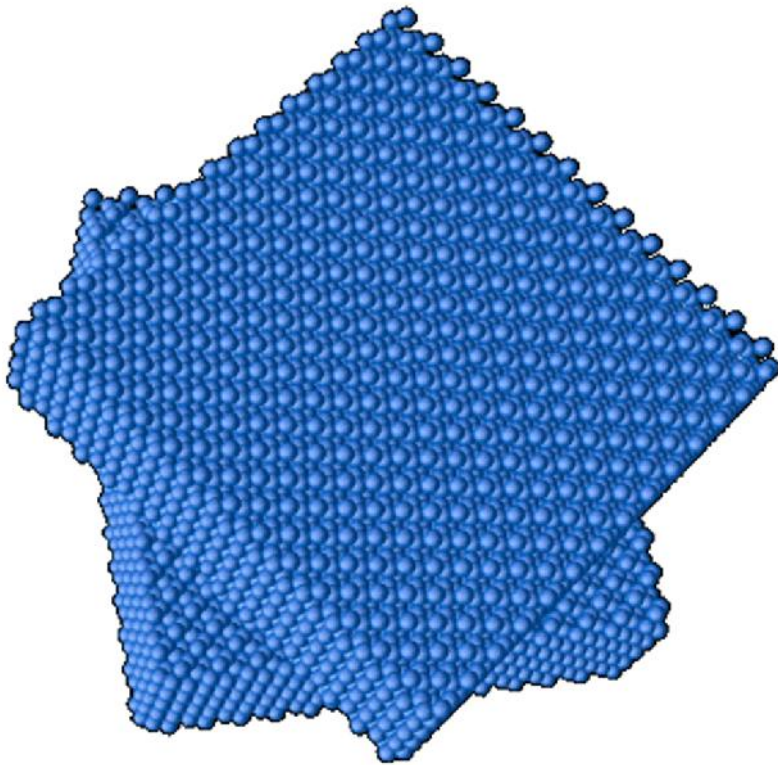


$$\{L\} = [L_1 \quad L_2 \quad L_3 \quad 0 \quad 0 \quad 0]^T$$

$$L_j = M_{jj}(\sigma_{+j} - \sigma_{-j}) \quad j = 1..3$$

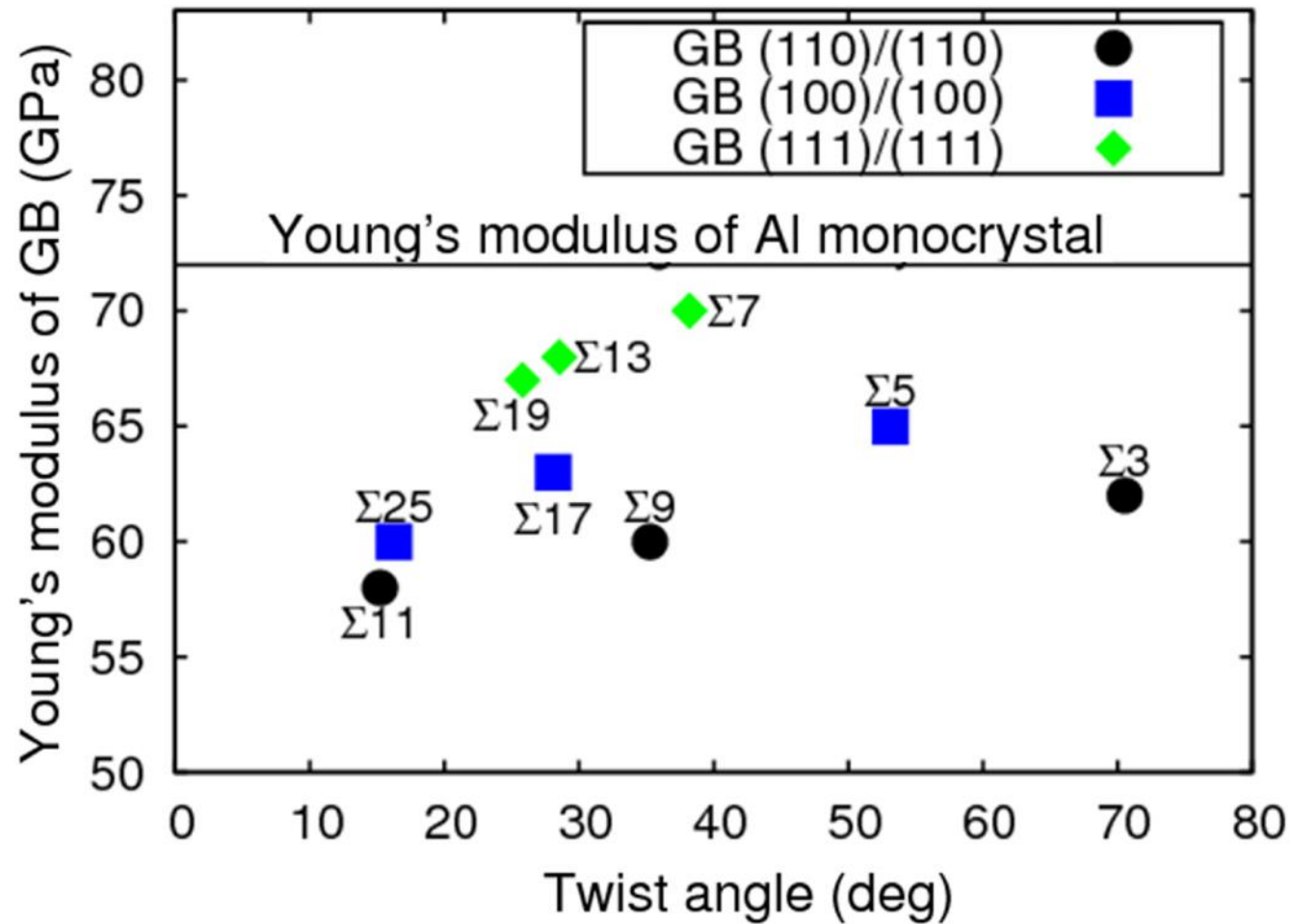
Young modulus of twist grain boundaries in Al

- 120-312 atoms in supercell
- Strain range from -3% up to 3%
- VASP code, GGA-PBE pseudopotential



Twist angle	Young modulus [GPa]
0	92
12.55	72
22.62	76
36.87	81

Young's modulus of grain boundary



Shear stress, shear modulus

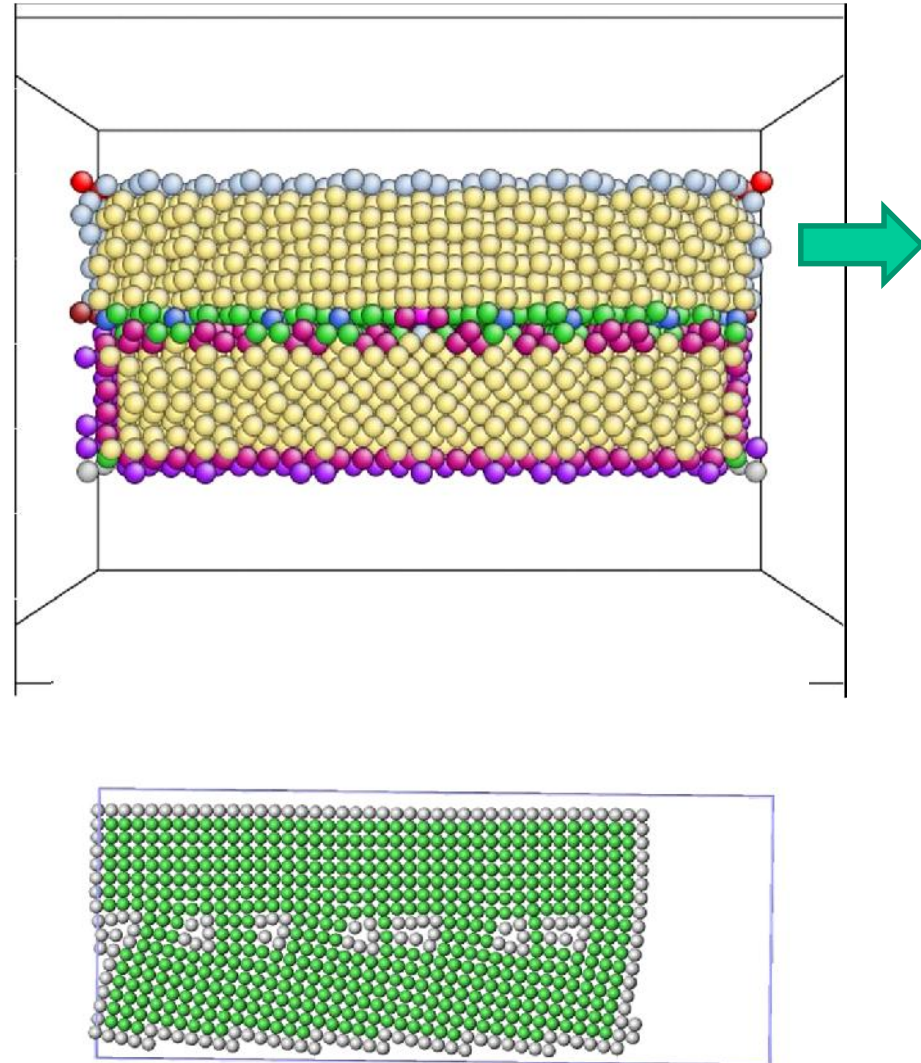
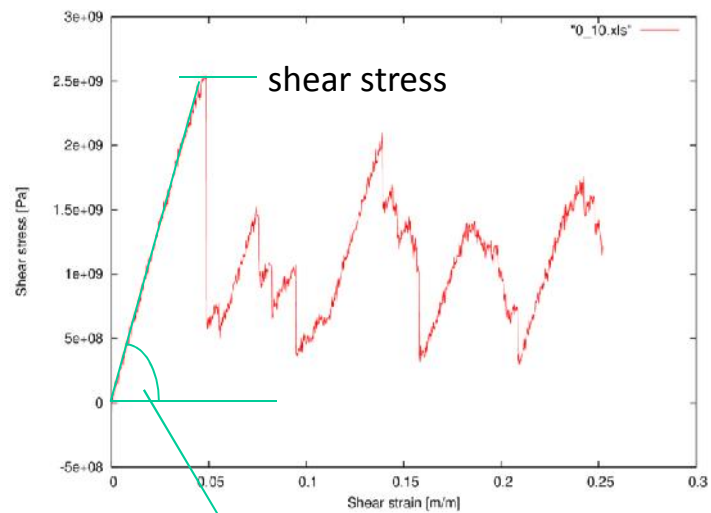
Effect of:

Structure

- Tilt, twist angle
- GB surface
- Size

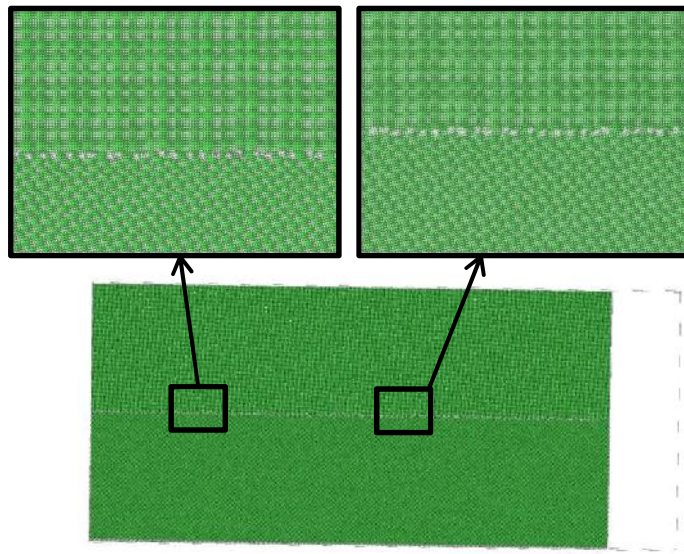
External conditions

- Strain rate
- Temperature



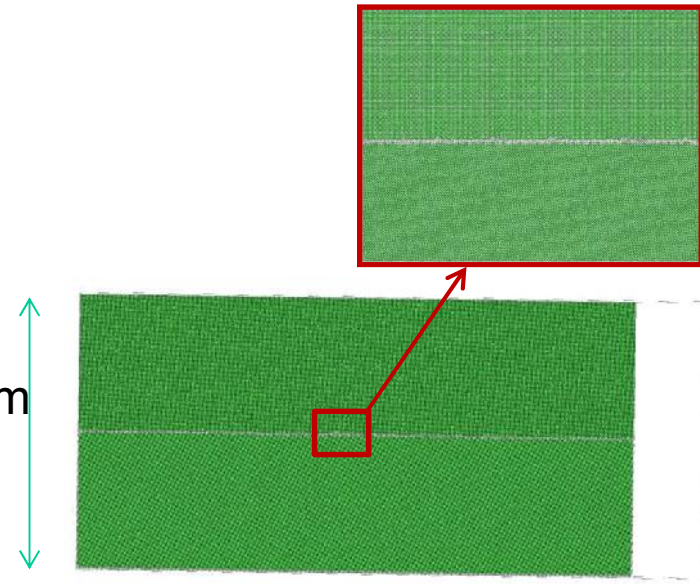
Shear stress, shear modulus

GB structure effect



Low-angle GB

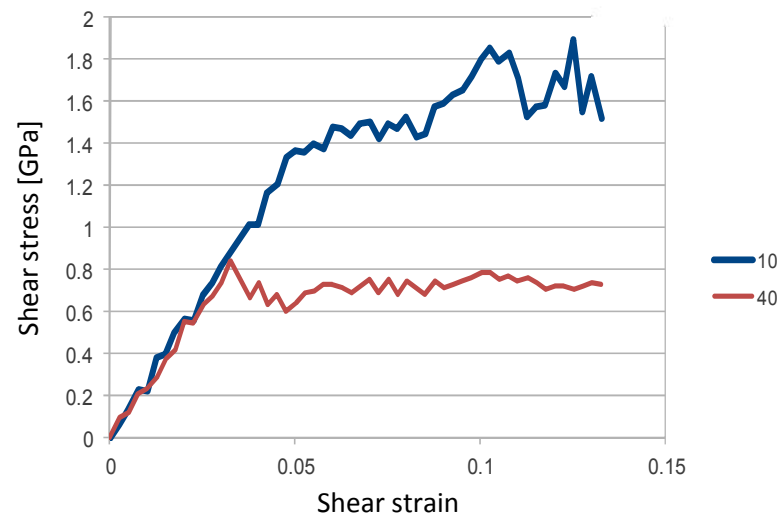
64 nm



High-angle (general) GB

Atomic scale

Molecular
Dynamics

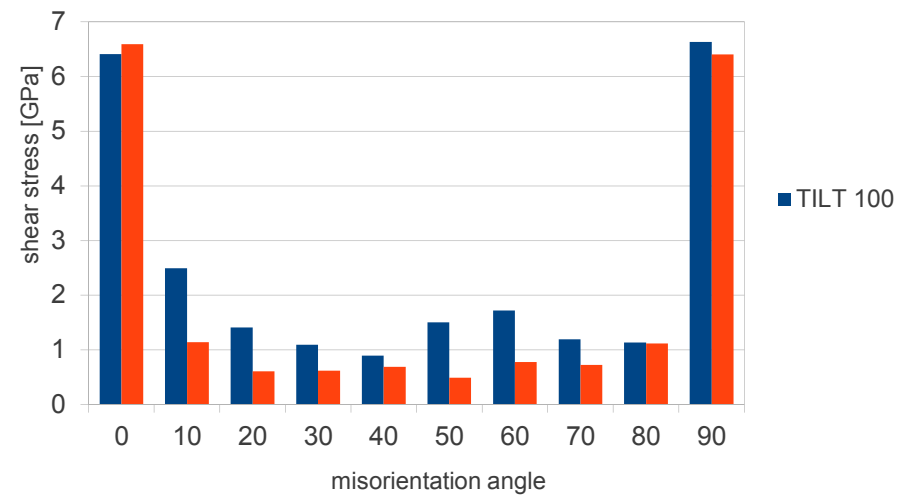


Shear stress, shear modulus

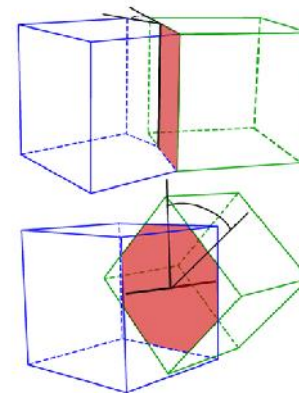
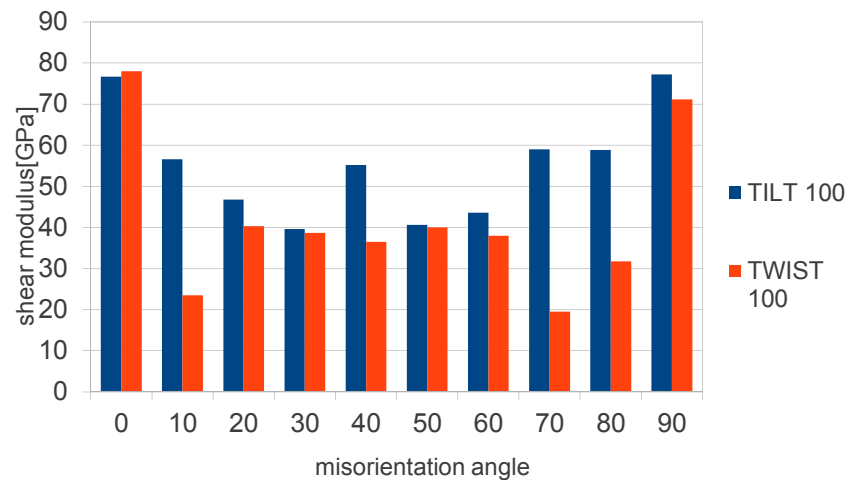
GB structure effect

Tilt and Twist angle

Misorientation angle vs shear stress

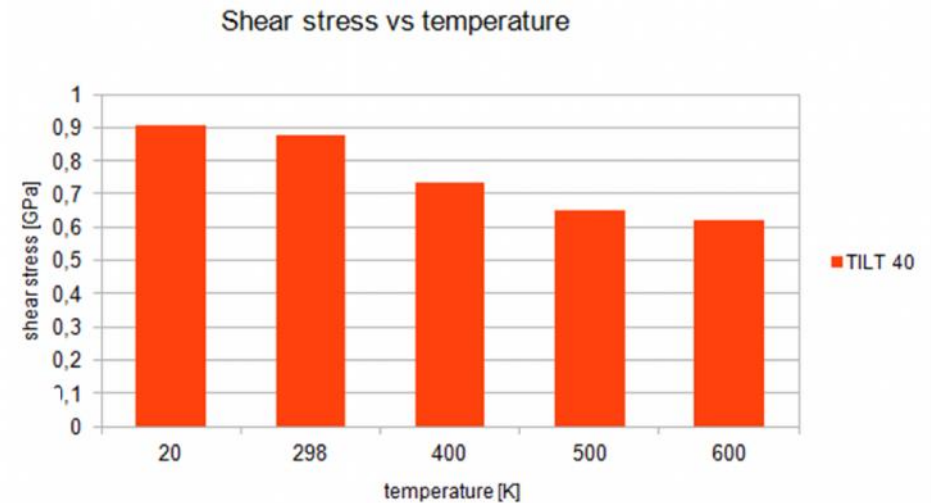
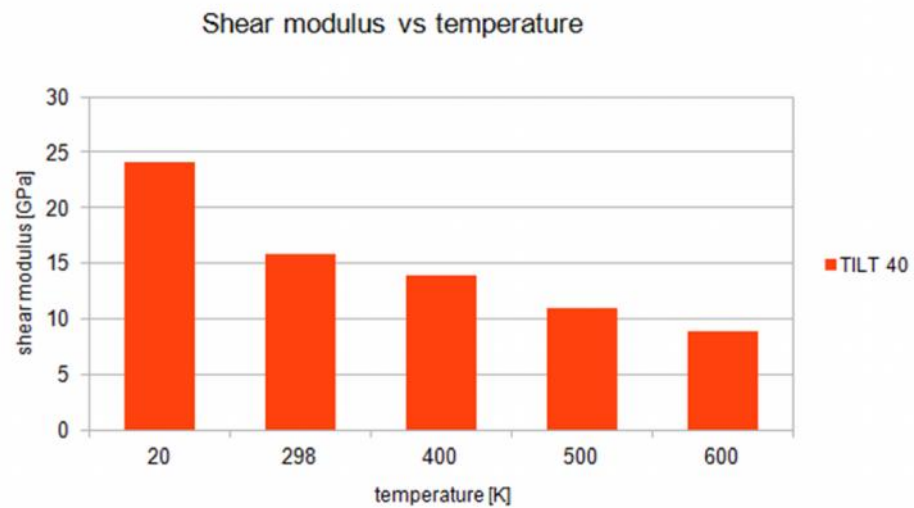


Misorientation angle vs shear modulus



Shear stress, shear modulus

Temperature effect



100

Macroscopic yield stress

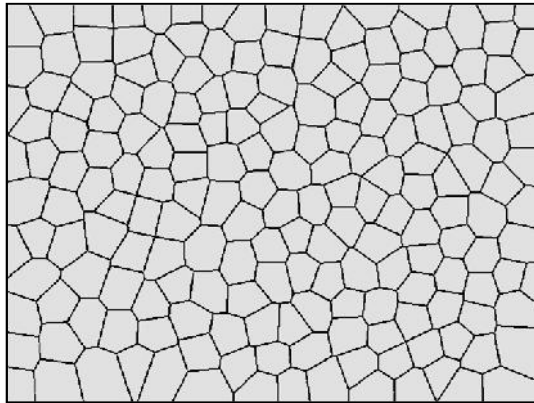
Variation coefficient

$$C_v = SD/E$$

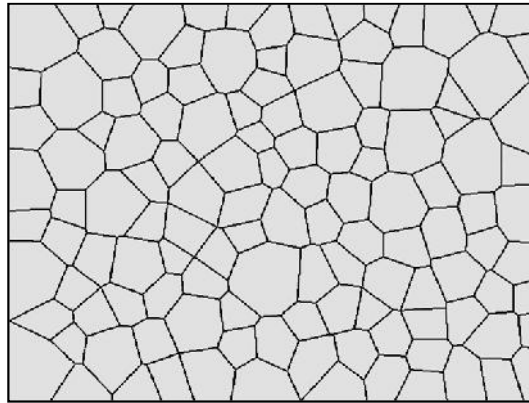
E - mean value

SD - standard deviation

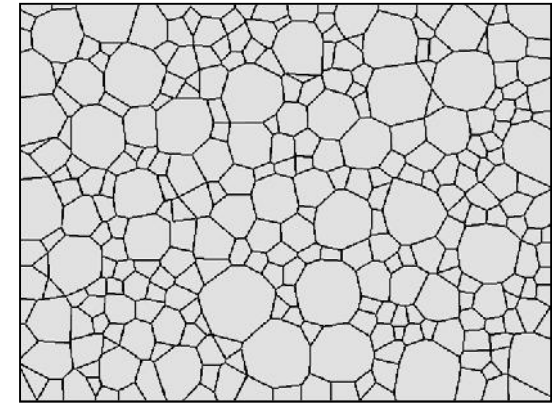
$C_v = 0.07$



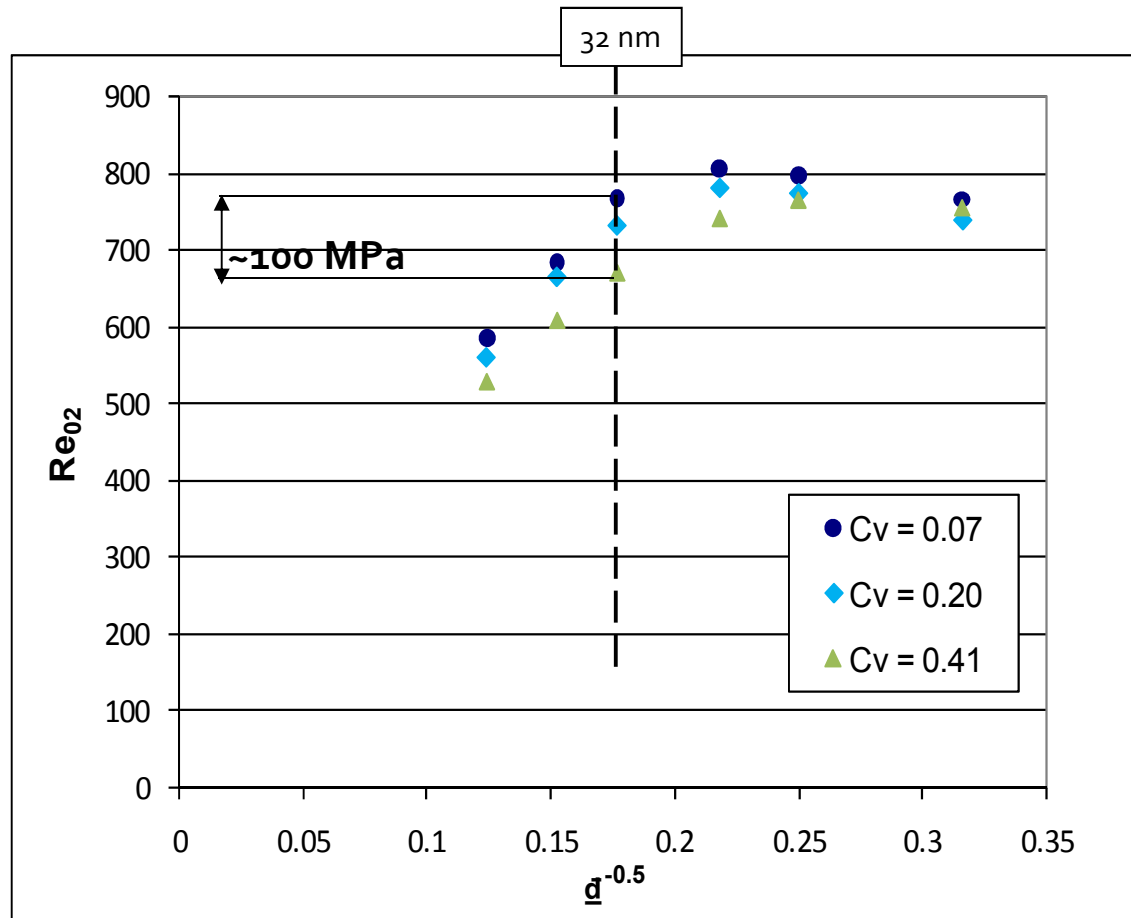
$C_v = 0.20$



$C_v = 0.41$



Macroscopic yield stress



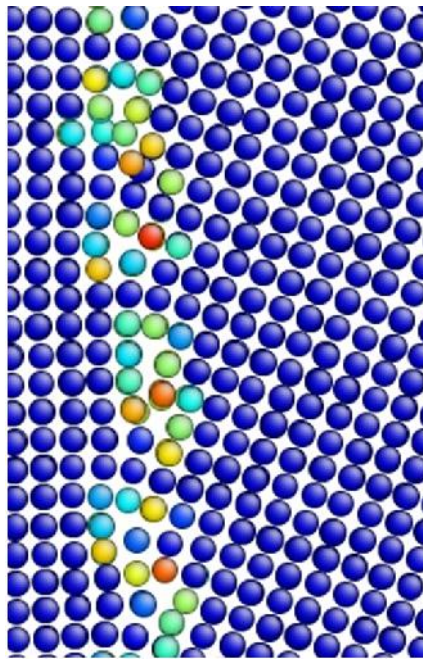
$C_v = 0.07$

$C_v = 0.20$

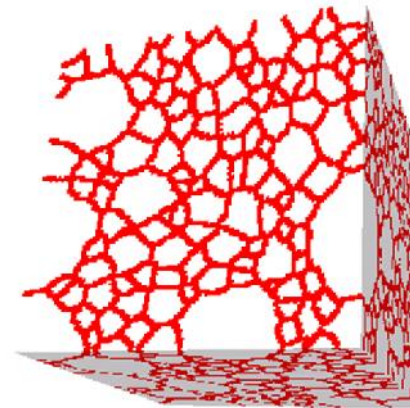
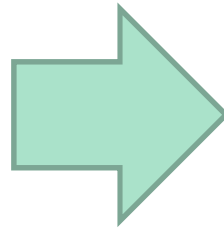
$C_v = 0.41$

Thermal stability

Grain boundary motion -> Grain Growth



Molecular dynamics with EAM
Atomic scale

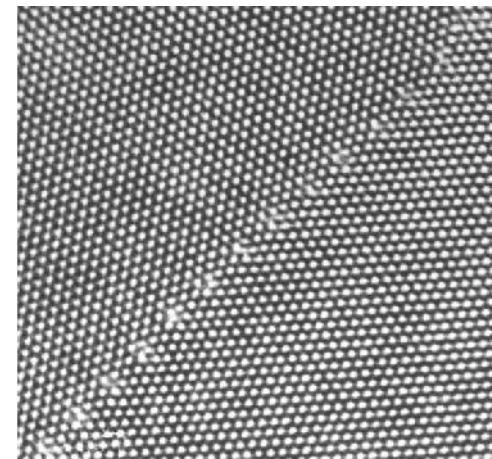
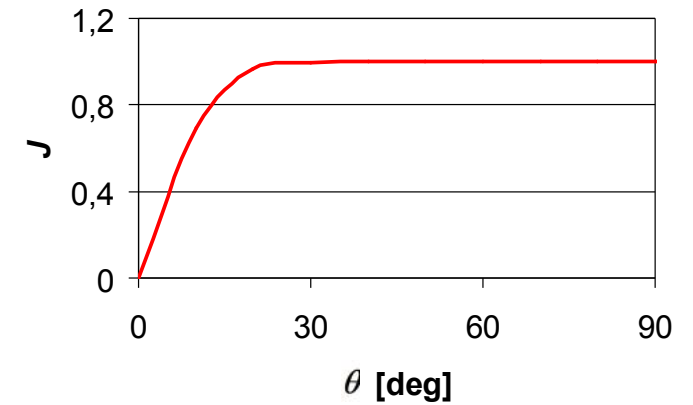
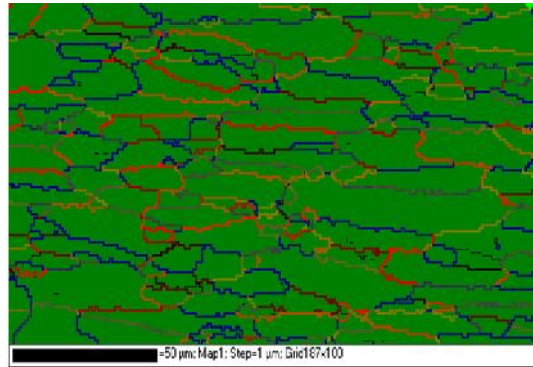
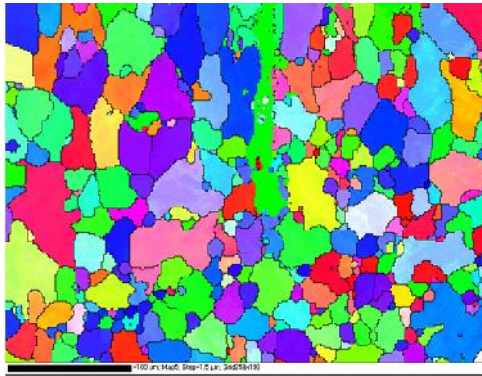


Monte Carlo model
Mesoscale

Grain growth mechanisms

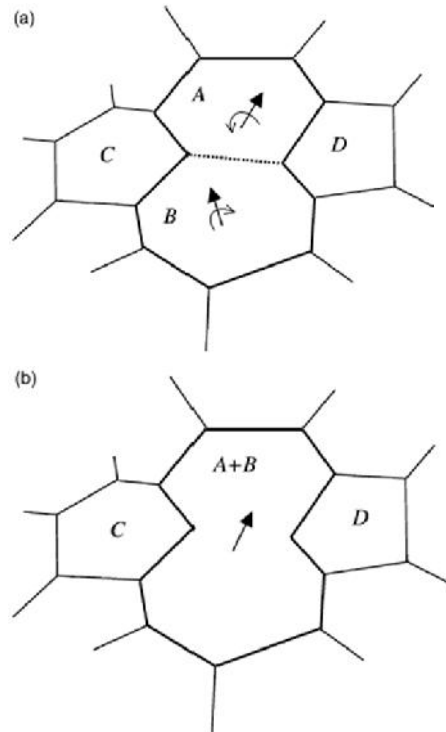
(disslocation based model *)

$$J(\theta) = J_{\max} \sin(2\theta) \{1 - A \ln[\sin(2\theta)]\}$$

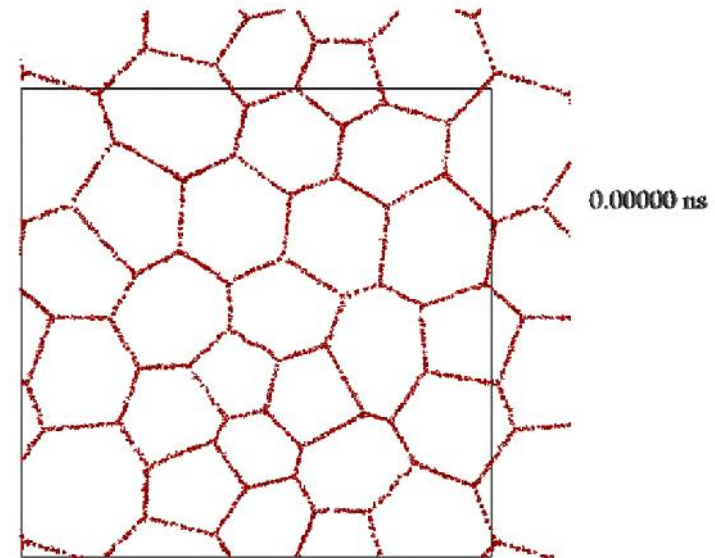
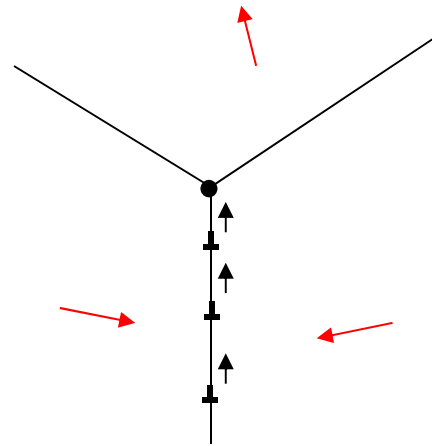


* W.T. Read, W. Shockley, Phys. Rev. 78 (1950) 275.

Grain growth mechanisms



Rotation of grains [1]



Pd, ~15 nm, 1400K [2]

$$p_o = A \left(1 - \frac{D_1}{D_0} \right) \left(1 - \exp \left[-B \left(\frac{\theta}{\theta_{\max}} \right)^n \right] \right)$$

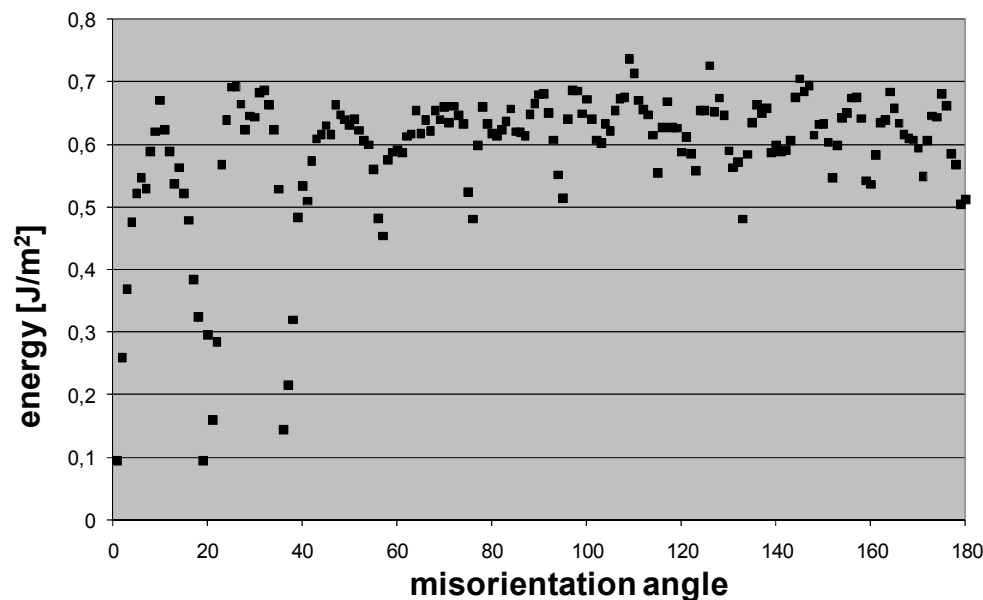
[1] D. Moldovan, D. Wolf a, S.R. Phillpot, A.J. Haslam, Role of grain rotation during grain growth in a columnar microstructure by mesoscale simulation, *Acta Materialia* 50, 3397–3414, 2002

[2] A.J. Haslam, S.R. Phillpot, D. Wolf, D. Moldovan, H. Gleiter, Mechanisms of grain growth in nanocrystalline fcc metals by molecular-dynamics simulation, *Materials Science and Engineering A*, 318, 1-2, 293, 2001

Grain boundaries – modelling of the energy

Results

Misorientation angle -> energy



$$\theta = \arccos\left(\frac{m_{11} + m_{22} + m_{33} - 1}{2}\right)$$

(100)

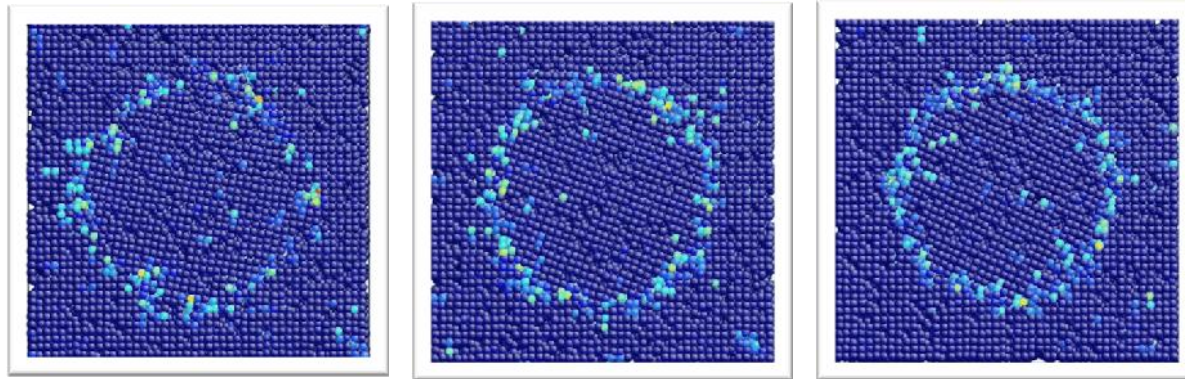
Good correlation
below 20 deg.

G. Zhu, W. Mao, Y. Yu, Calculation of Misorientation distribution between Recrystallized Grains and Deformed Matrix, Scripta mater. 42, 37, 2000.

Thermal stability

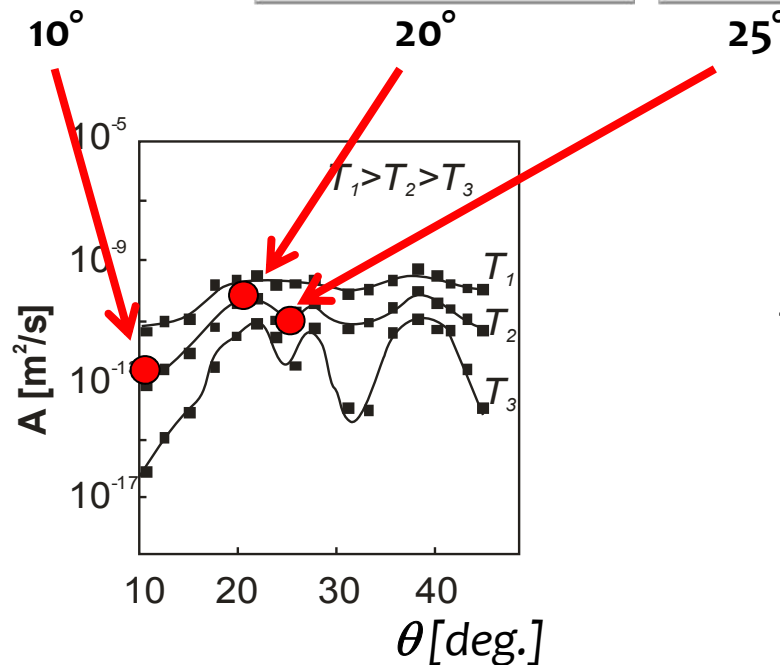
Atomistic scale - Molecular Dynamics

Grain boundary mobility vs misorientation and curvature



Nanocrystalline iron –
1000K

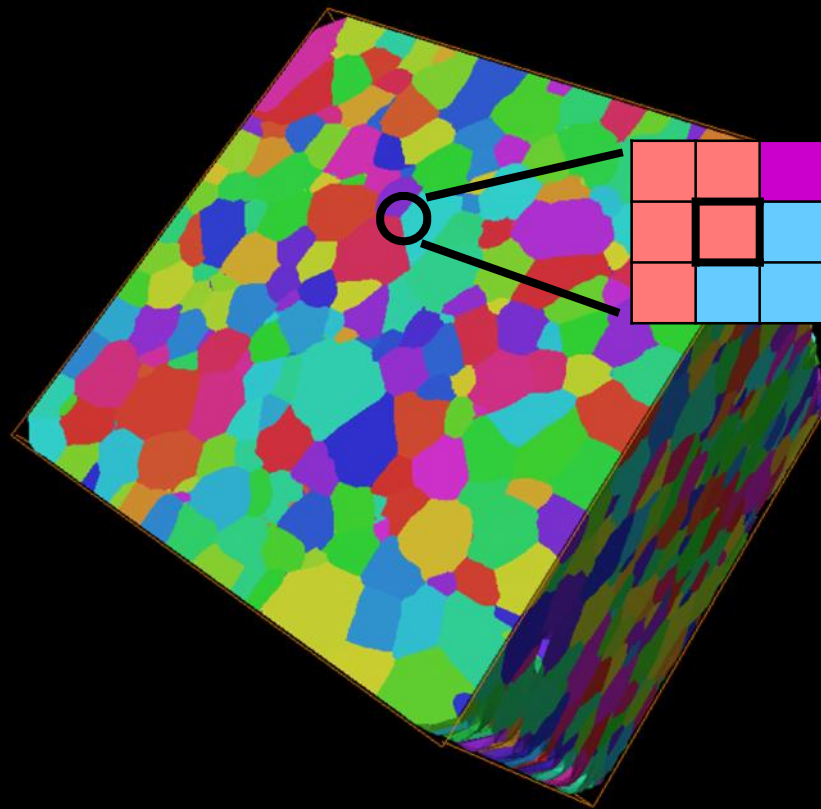
Size: 9nm, time: 2ns



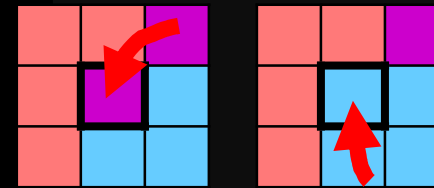
T. Wejrzanowski, M. Spychalski, R. Pielaszek,
K.J. Kurzydowski, *Grain boundary migration in
nanocrystalline iron*, Solid State Phenomena 129,
2007, 145-150

3D model

Monte Carlo



New configuration



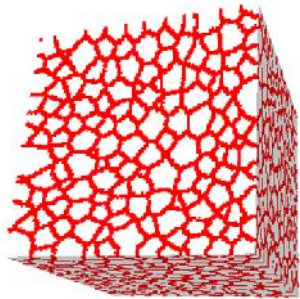
$$G = - \sum_{ij} J_{ij} (\delta_{ij} - 1)$$

$$p = \begin{cases} \exp(-\Delta G / k_b T), & \text{dla } \Delta G > 0 \\ 1, & \text{dla } \Delta G \leq 0 \end{cases}$$

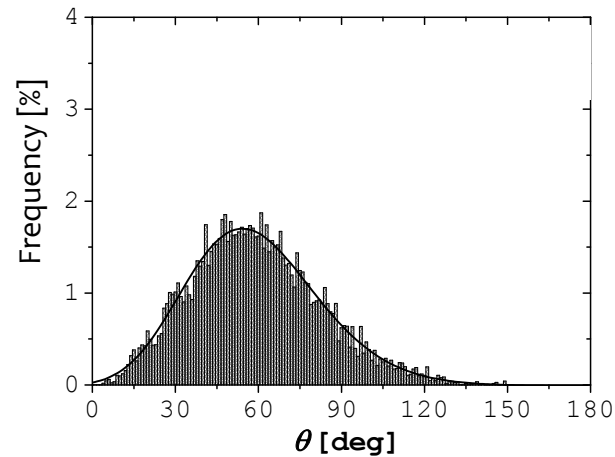
Grain size inhomogeneity effect

Monte Carlo

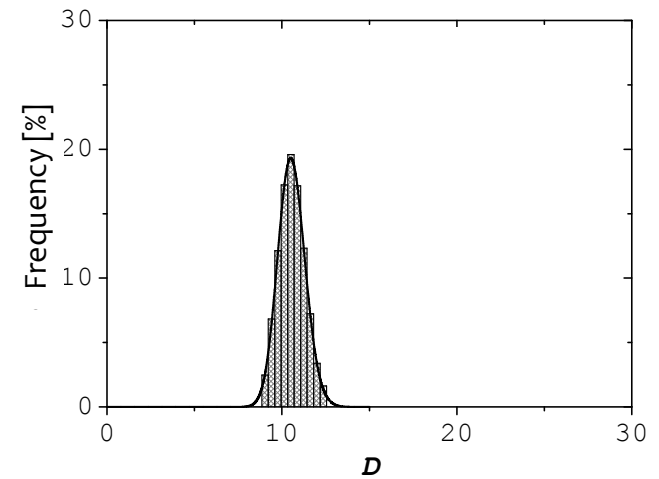
$CV(D)=0,13$



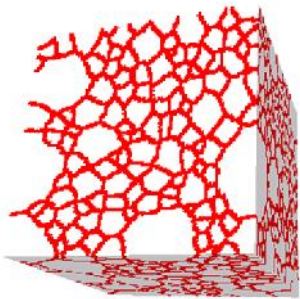
homogenous



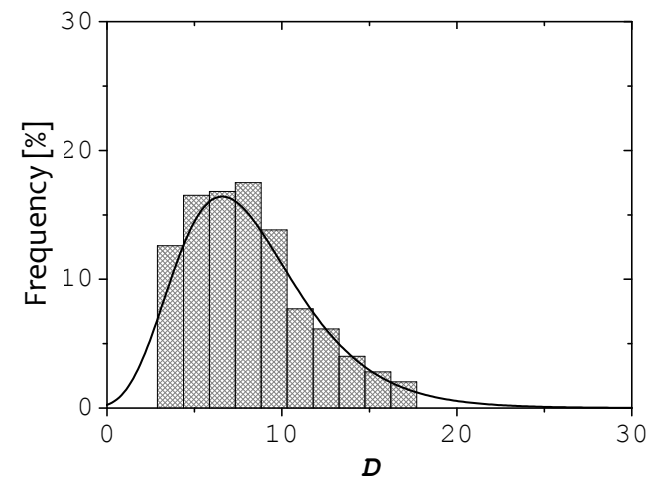
Grain size distribution



Non-homogenous



GB misorientation distribution



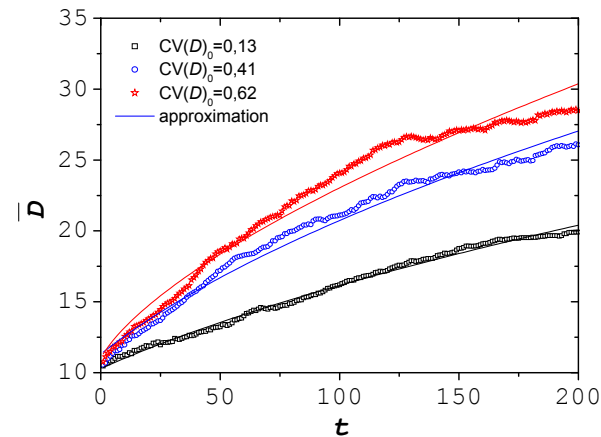
$CV(D)=0,62$

Thermal stability

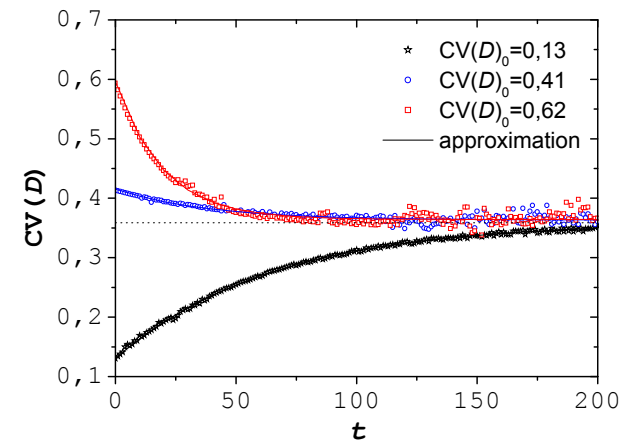
Grain growth in nanometals

Mesoscale - MonteCarlo

Influence of grain size homogeneity



Changes of grain size



$$CV(D) = 0,36 + A \exp(-Bt)$$

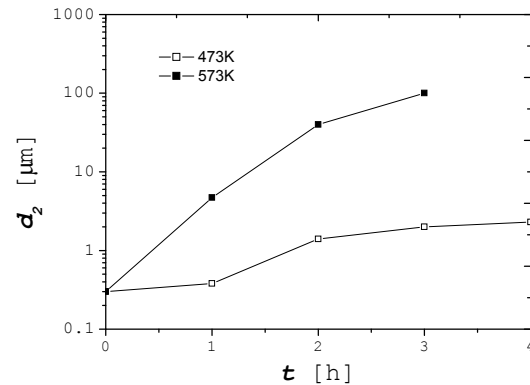
Changes of grain size dispersion

M. Lewandowska, T. Wejrzanowski, K.J. Kurzydowski, Grain growth in ultrafine grained aluminium processed by hydrostatic extrusion, Journal of Materials Science, 43, 2008, 7495–7500

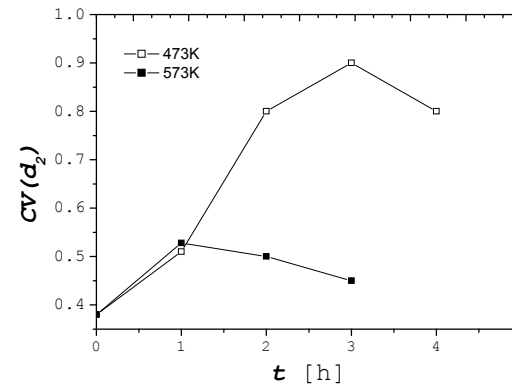
Thermal stability

Experimental verification – time effect

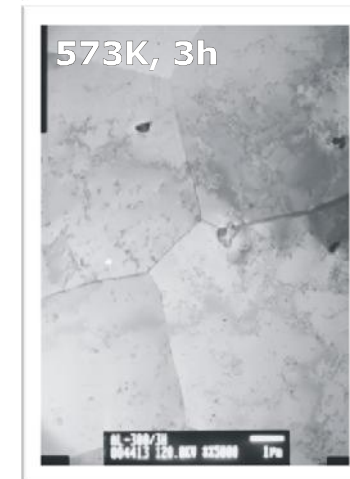
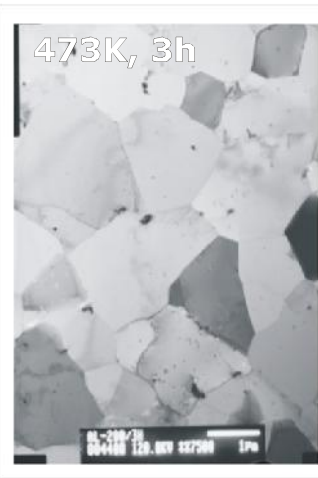
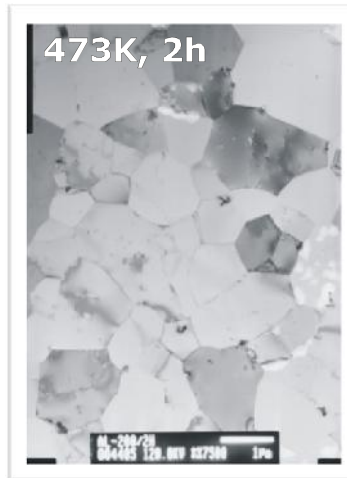
constant temperature



Kinetics of grain growth in aluminium after Hydroextrusion annealed at defined temperature and different time



Temporal changes of grain size dispersion in samples annealed at different temperature



Melting point

Boundary condition effect

Lindemann index - root mean squared (rms) distance fluctuation

$$\delta = \frac{2}{N(N-1)} \sum_{i < j} \frac{\sqrt{\langle r_{ij}^2 \rangle - \langle r_{ij} \rangle^2}}{\langle r_{ij} \rangle}$$

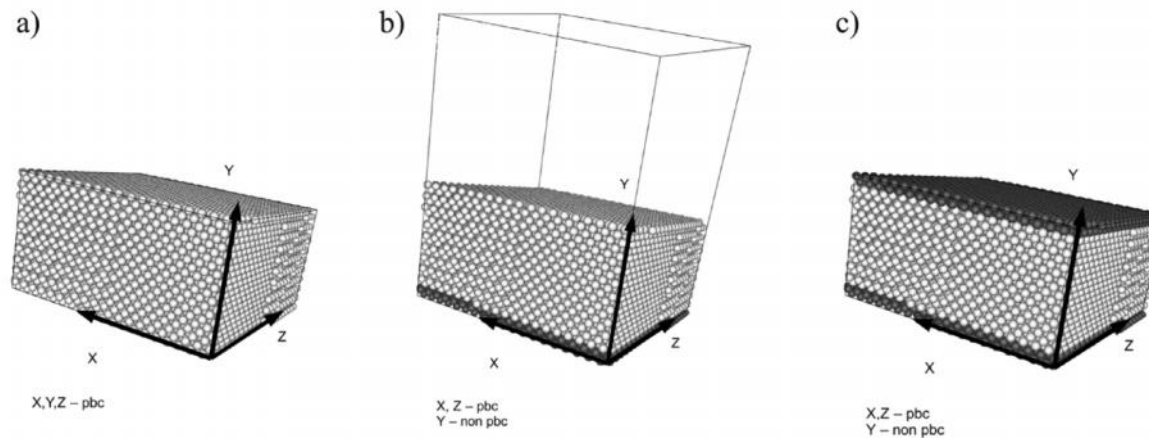


FIG. 1. Schematic illustration of the systems modeled in this study: (a) infinite crystal, (b) with a free surface, (c) with a thin interlayer.

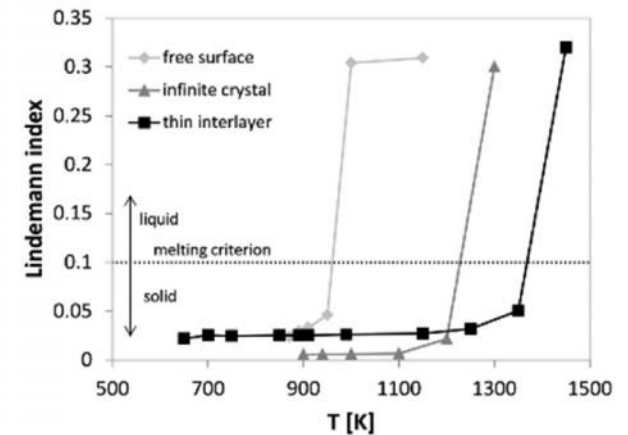


FIG. 3. The effect of boundary conditions on the Lindemann index for aluminum thin film.

T. Wejrzanowski, M. Lewandowska, K. Sikorski, K. J. Kurzydowski,
Effect of grain size on the melting point of confined thin
aluminum films, Journal of Applied Physics 116, 164302 (2014)

Melting point

Time effect

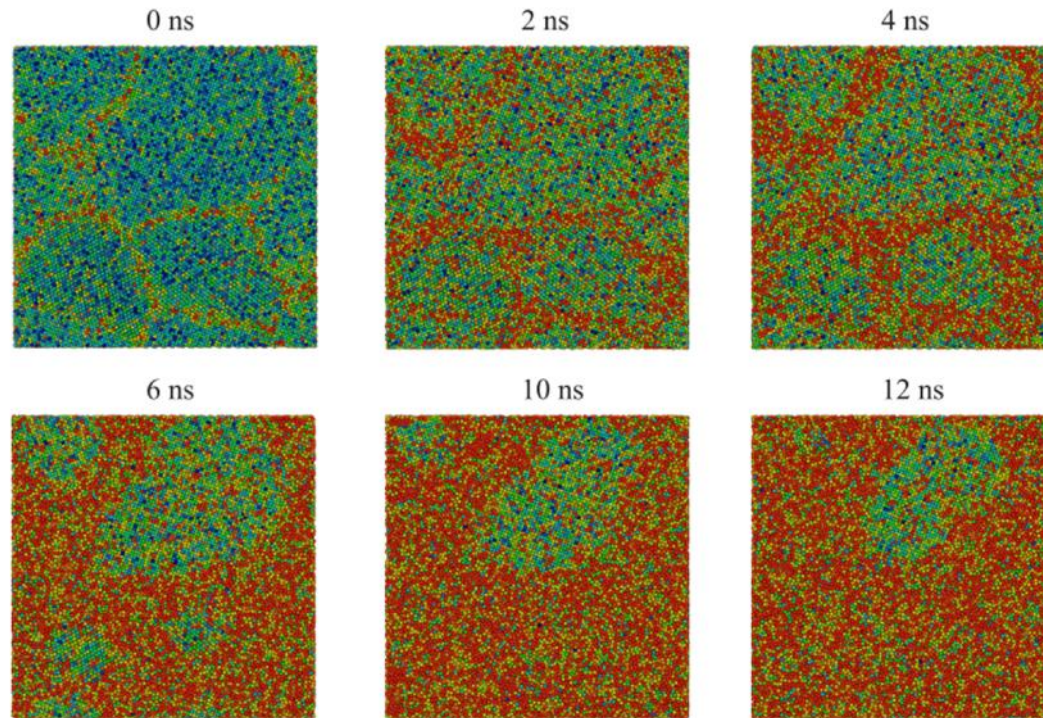


FIG. 5. Evolution of the system with average grain diameter of 11 nm at 1000 K. Red indicates the highest energy and blue the lowest.

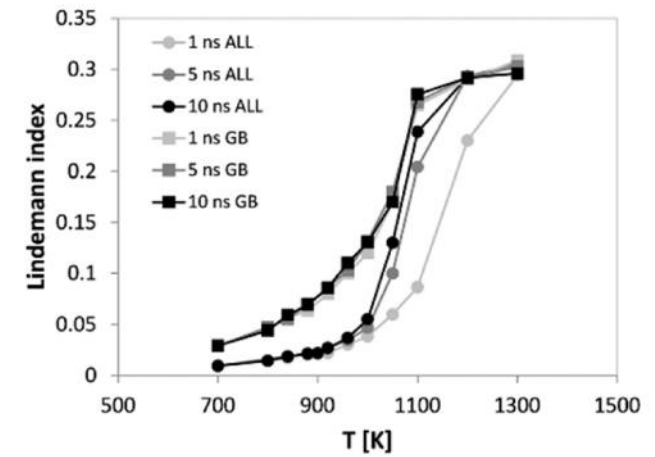


FIG. 6. Lindemann index calculated for all atoms and for atoms near grain boundaries after different annealing time.

Melting point

Grain size (GB surface) effect

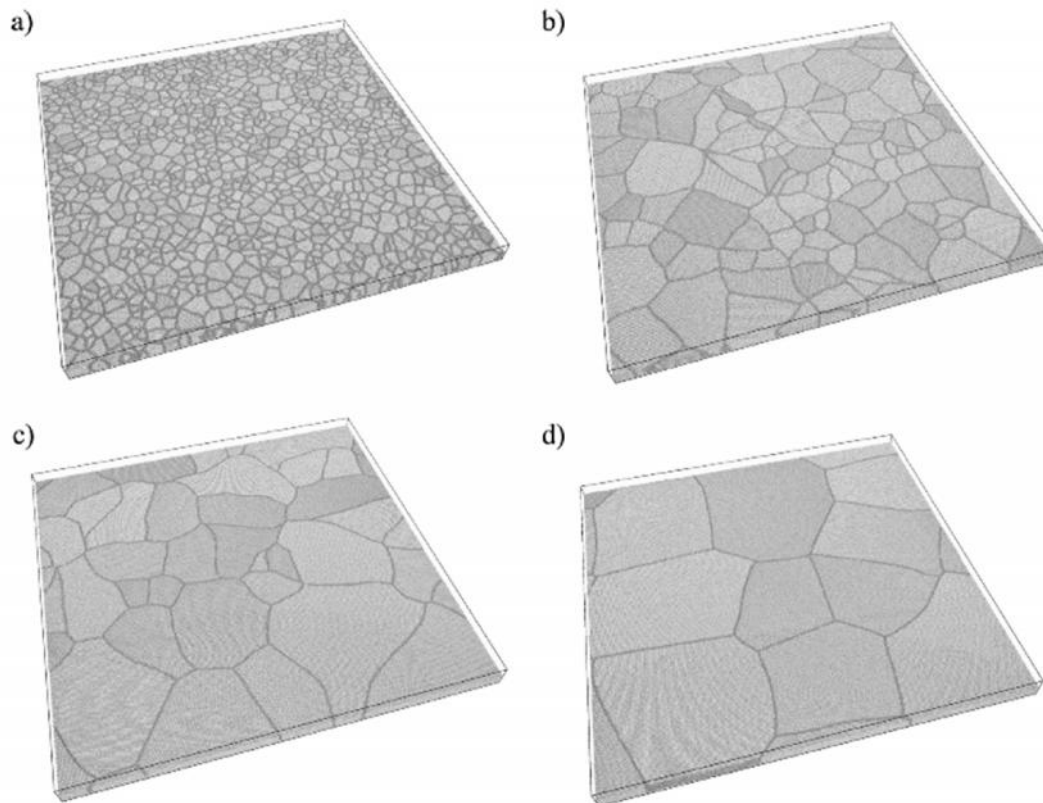


FIG. 2. Atomic structures for modeling the effect of grain size on melting point in the aluminum interlayer.

Size and pressure effect

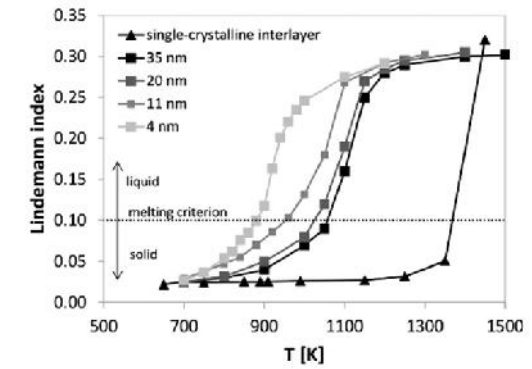


FIG. 7. Effect of grain size on temperature related Lindemann index.

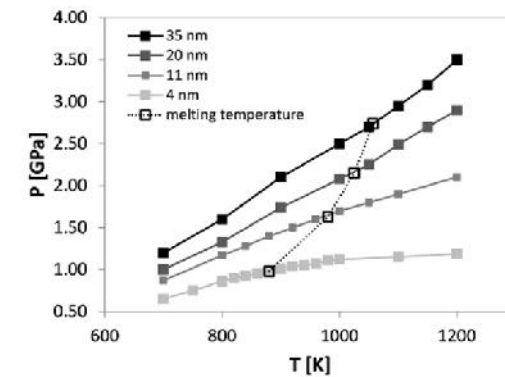


FIG. 10. Pressure vs temperature for the confined film with different grain

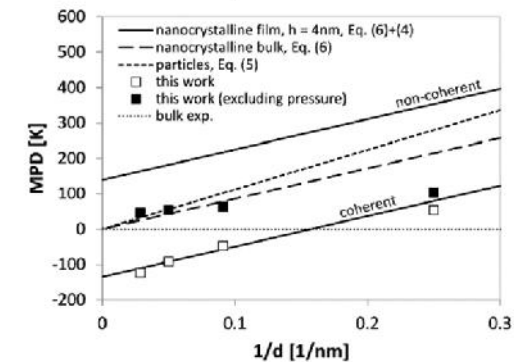


FIG. 9. Effect of grain size on MPD for particles and nanocrystalline bulk and films with different type of confinement.

Corrosion resistance

Effect of grain boundary energy on intergranular corrosion

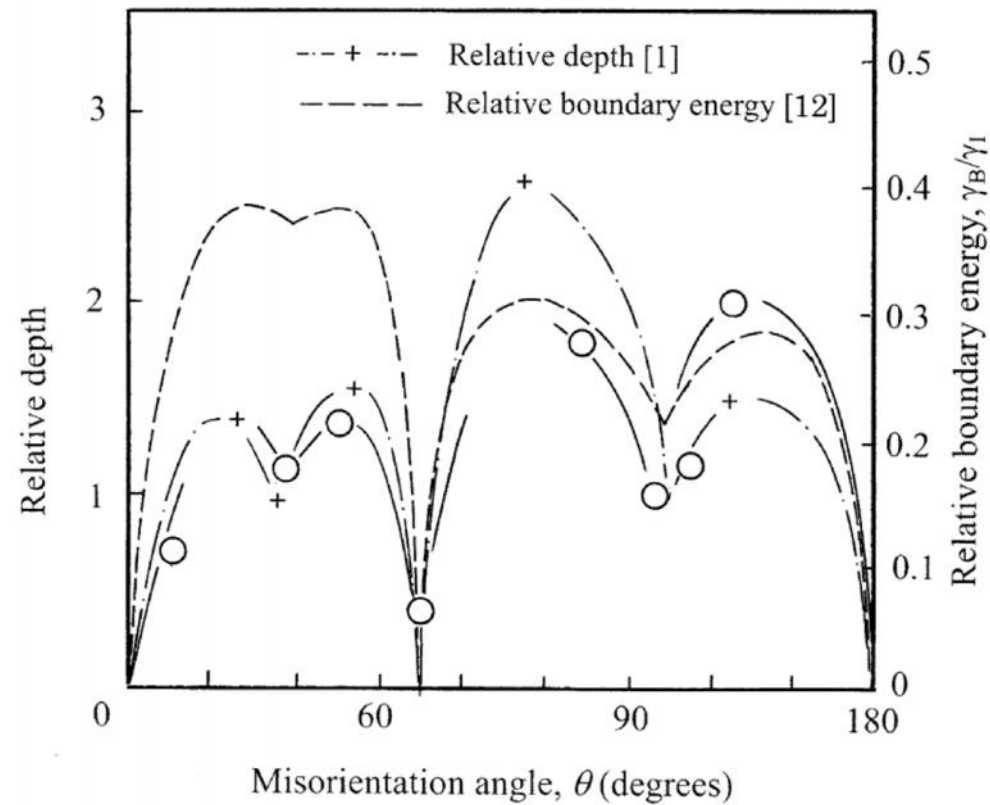
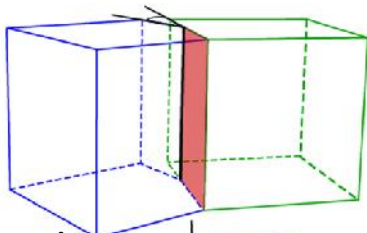


Fig. 5. Misorientation dependences of the relative depth of intergranular grooves and relative grain boundary energy reported by Miura et al. [12]. Relative depth of intergranular grooves in [1 1 0] tilt pure copper bicrystals reported by Yamashita et al. [1] is also shown.

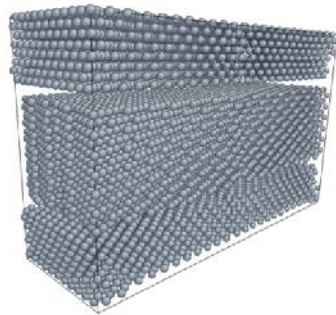
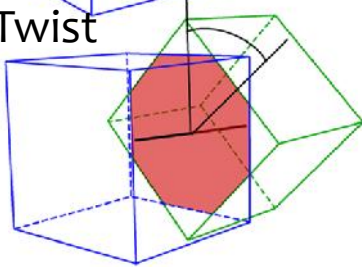
Corrosion resistance

GB energy vs misorientation

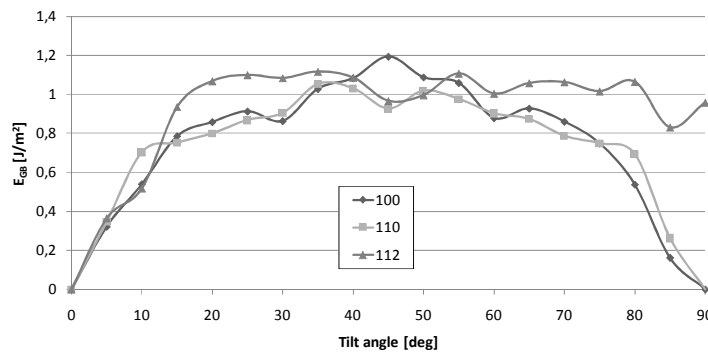
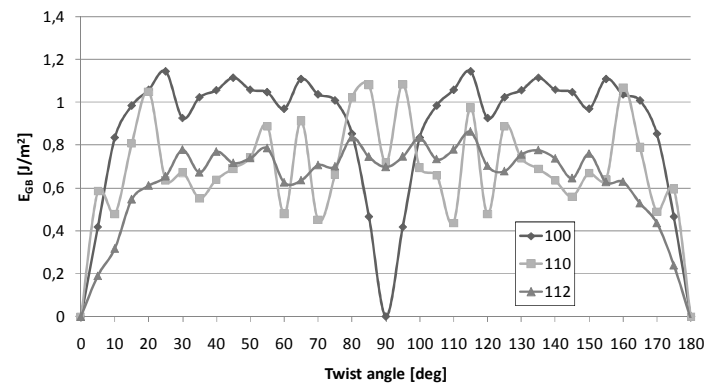
Tilt



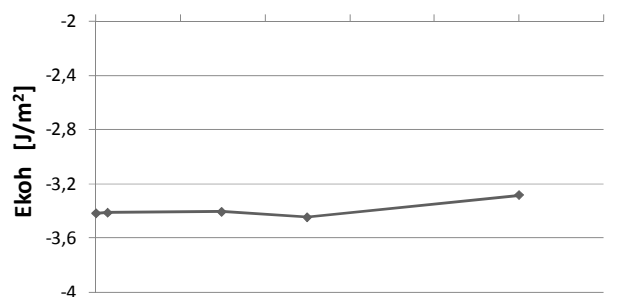
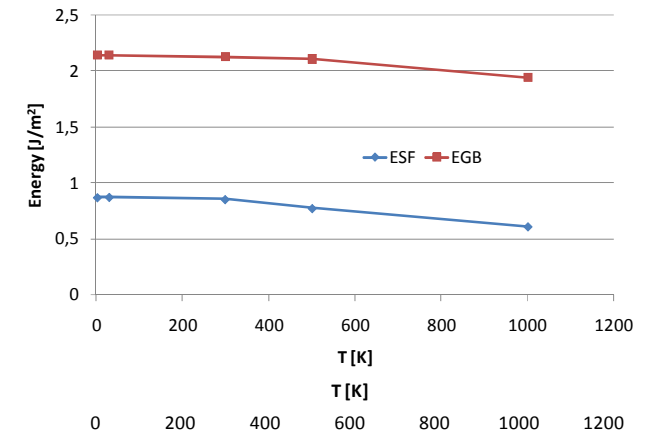
Twist



Surface and misorientation
(room temperature, atmospheric pressure)

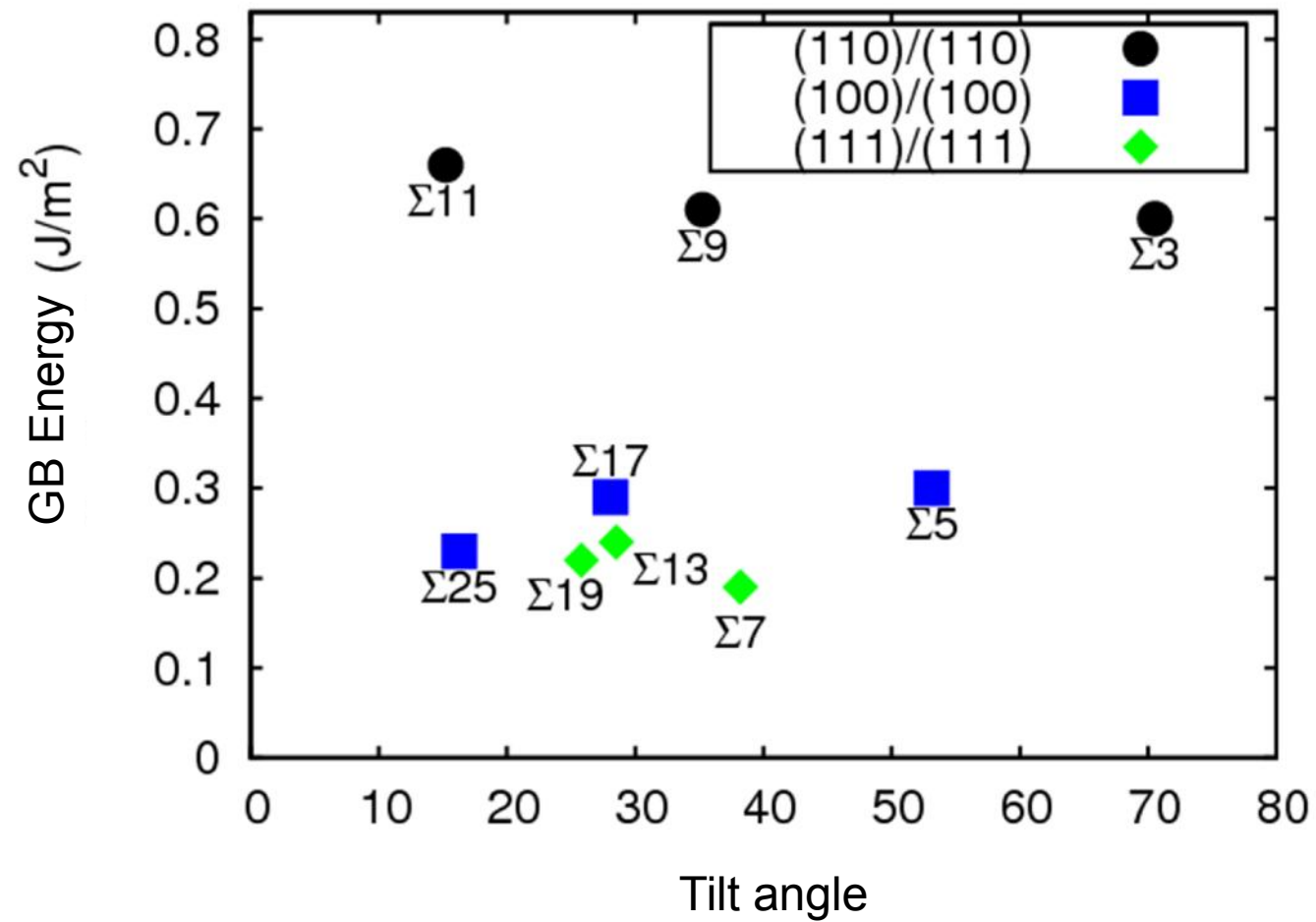


Temperature behaviour of 210 Fe GB



ESF – free surface energy
EGB – grain boundary energy
E_{koh} – cohesion energy

Grain boundary energy in Al



Corrosion resistance

Grain boundaries in aluminium alloys

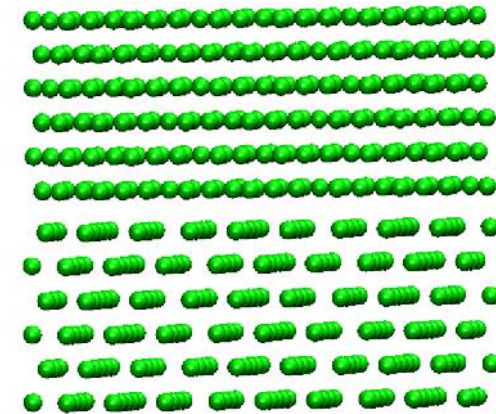
Fields of interest:

- Grain boundary energy
- Structure of grain boundary
- Segregation energy of impurities
- Cohesion of grain boundary

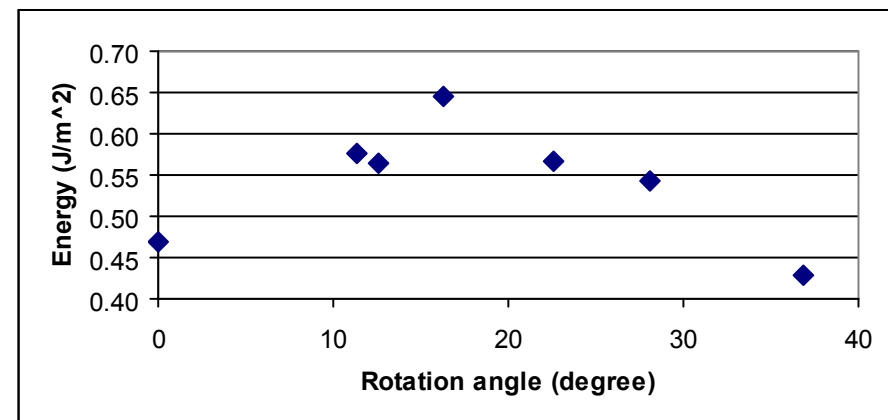
Segregation energy of impurities in grain boundary in Al

Angle	Cu	Fe	Mg	Mn
12.55	0.1596	0.1211	-0.0418	0.1234
22.62	-0.3510	0.1206	-0.0397	0.1330
36.87	-0.1700	0.1817	-0.0629	-0.1794

Atomic scale – ab initio



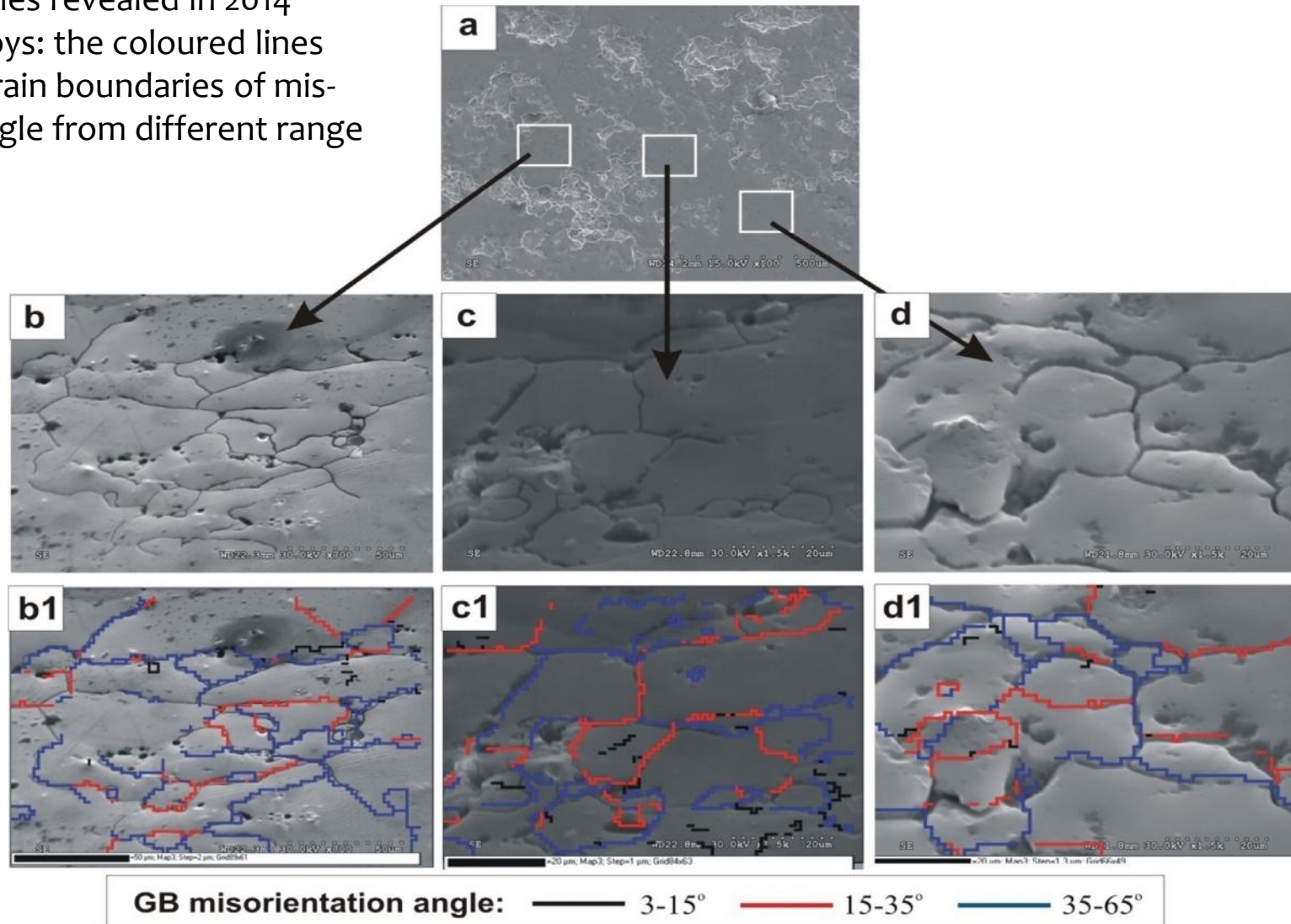
Positions of the atoms in aluminium calculated for a flat grain boundary



Corrosion resistance

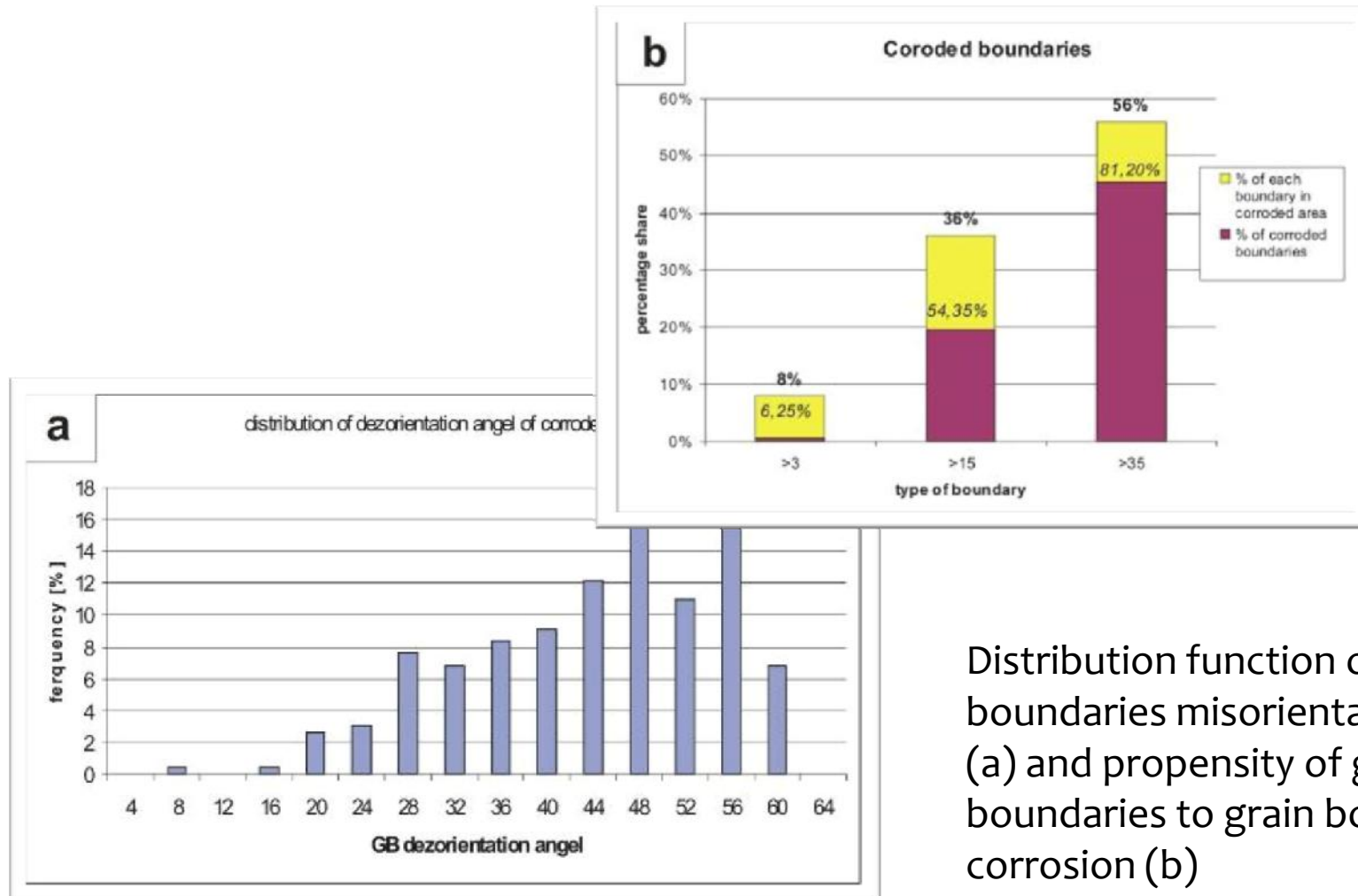
Experimental evidences

Grain boundaries revealed in 2014 aluminium alloys: the coloured lines indicate the grain boundaries of misorientation angle from different range



Corrosion resistance

Experimental evidences



Distribution function of the grain boundaries misorientation angle (a) and propensity of grain boundaries to grain boundary corrosion (b)

Graphene Composites - Electronics cooling



MATERIALS DESIGN DIVISION

- Limiting factor
- Densely packed integrated circuits
- More computational power

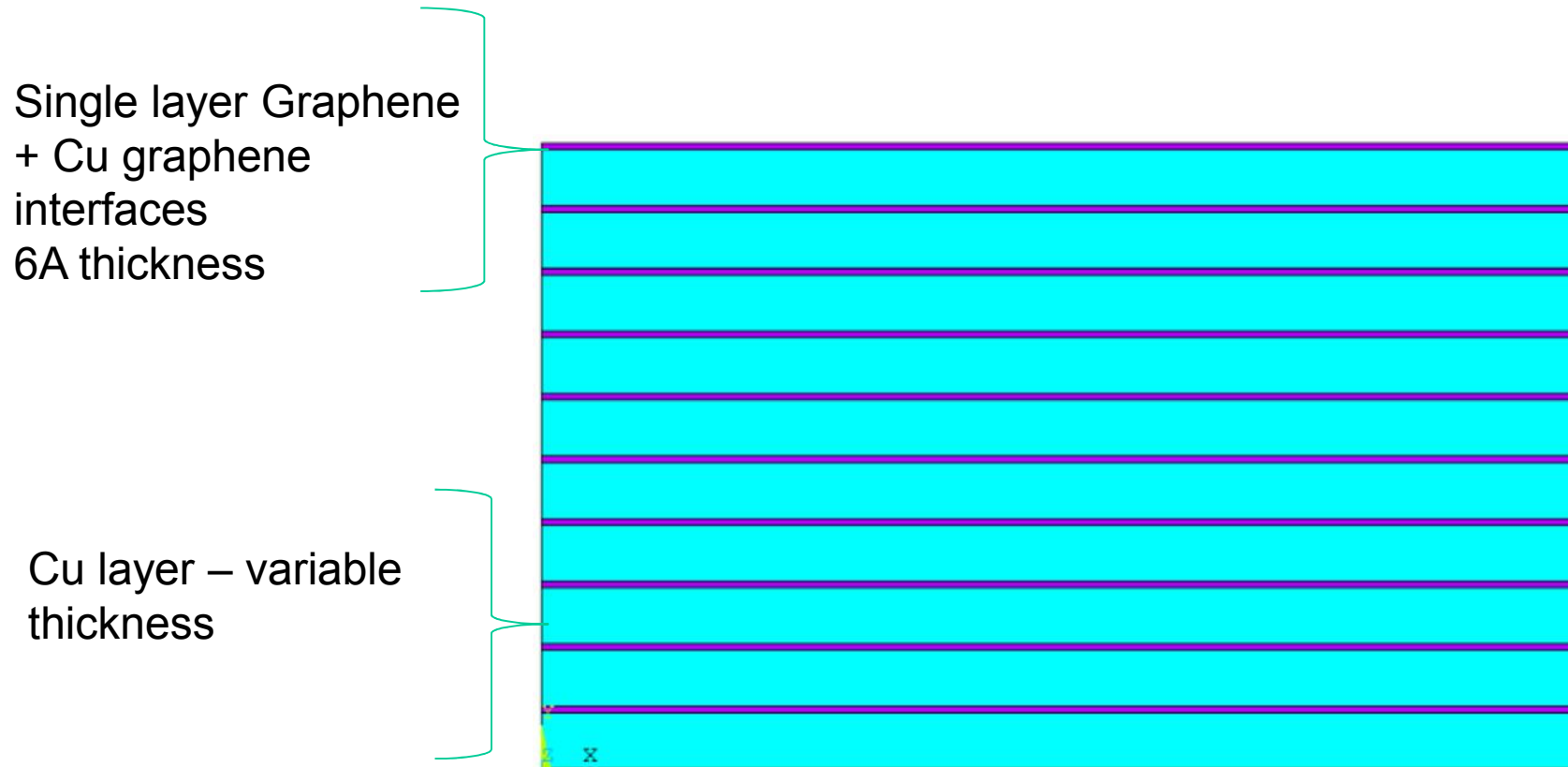


<http://www.computerworlduk.com/>

- Specific shape of electronic devices – flat, thin – heat flow is limited by small area of cooling elements and at the same time heat must travel long way to be dissipated

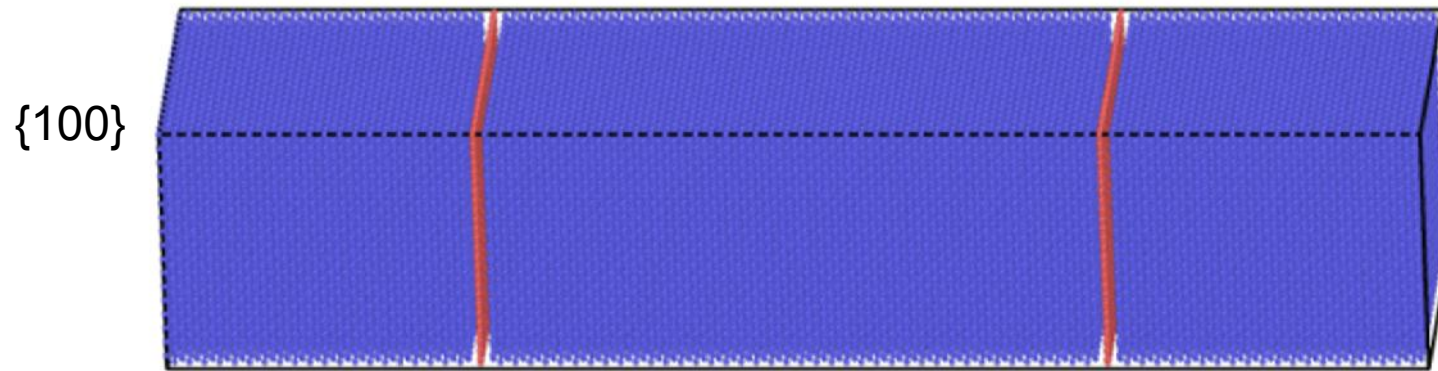
Moore A.L., Shi L., *Emerging challenges and materials for thermal management of electronics*. Materials Today. 2014, 17(4), 163-74.

Graphene – metal composite for energy dissipation from localized heat sources



Dimensions and properties rescaled because of FEM and computer number representation limitations

Atomic scale

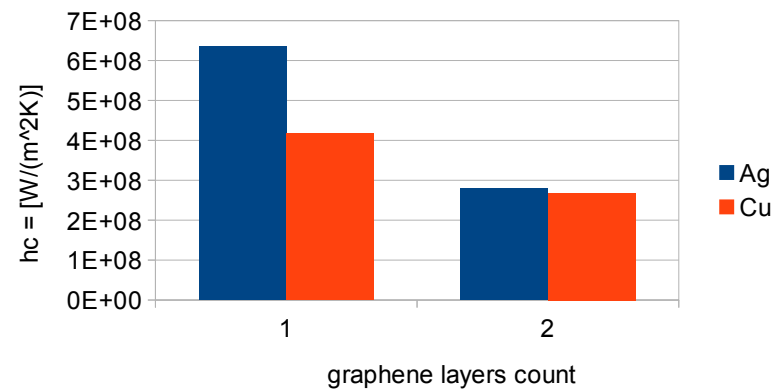


Periodic Boundary conditions

Atom count: 114240

Box dimensions: 62x72x303 Å

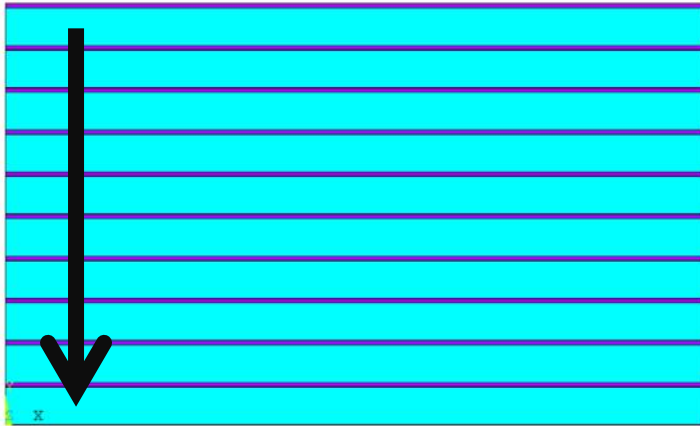
Time: ~150 ps





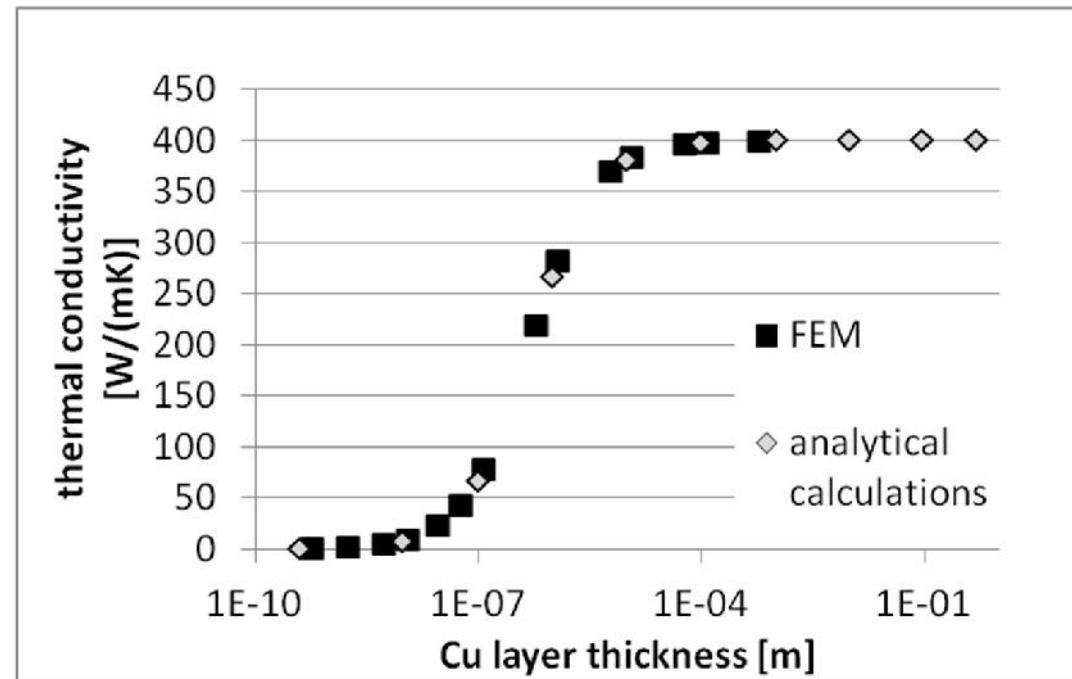
Results TC across layers

FEM and analytical calculations results



$$\mathbf{A} = \frac{\mathbf{E}}{\mathbf{R}_{\Omega}} \xleftrightarrow{\text{Analogously}} \dot{\mathbf{Q}} = \frac{\Delta \mathbf{T}}{\mathbf{R}_{th}}$$

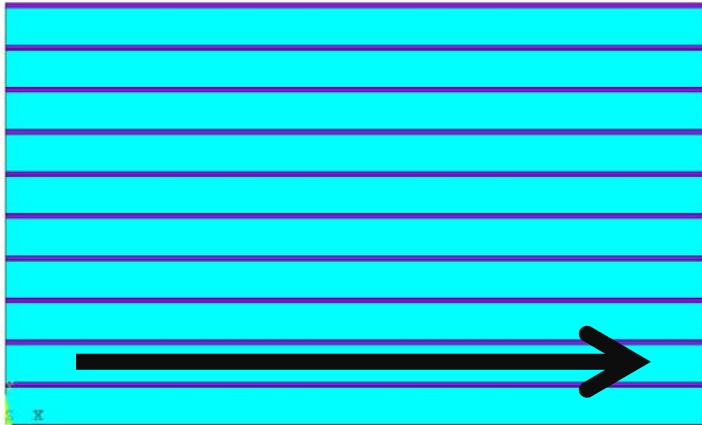
$$R_{th} = \frac{l}{A} R_{\lambda} = \frac{l}{A} \frac{1}{\lambda} \left[\frac{K}{W} \right] \quad R_{th} = \frac{1}{A h_c} \left[\frac{K}{W} \right]$$



$$R_{s \perp} = R_1 + R_2 + R_4 + \dots + R_{2n+1}$$

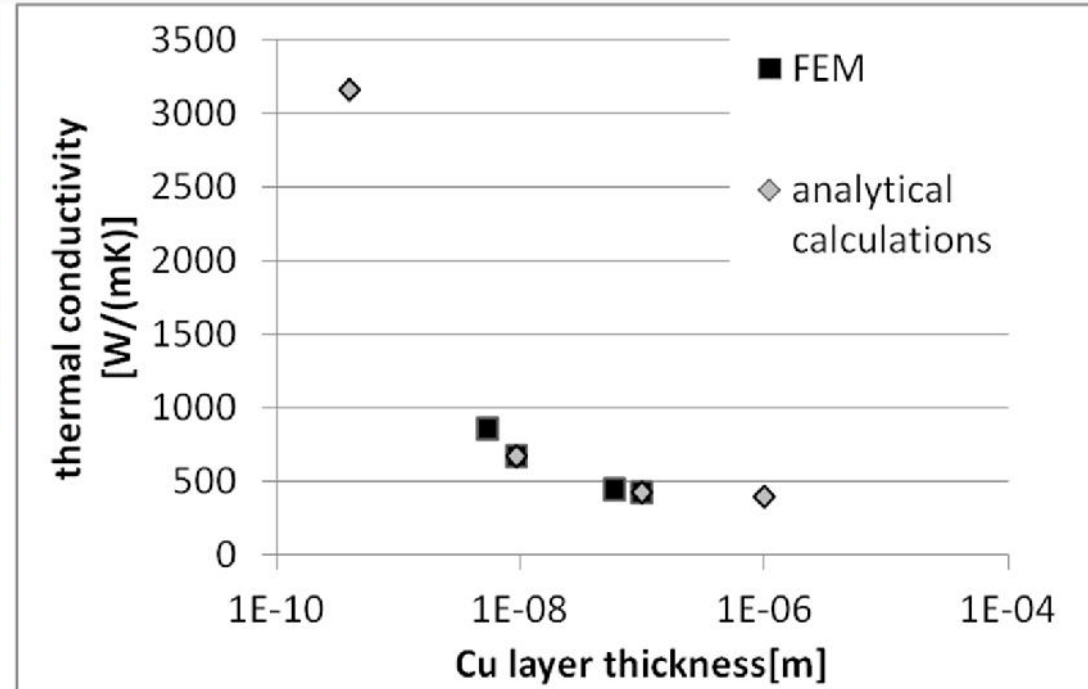
Results – TC along layers

FEM and analytical calculations results



$$\mathbf{A} = \frac{\mathbf{E}}{\mathbf{R}_{\Omega}} \xleftrightarrow{\text{Analogously}} \dot{\mathbf{Q}} = \frac{\Delta \mathbf{T}}{\mathbf{R}_{th}}$$

$$R_{th} = \frac{l}{A} R_{\lambda} = \frac{l}{A} \frac{1}{\lambda} \left[\frac{K}{W} \right] \quad R_{th} = \frac{1}{Ah_c} \left[\frac{K}{W} \right]$$

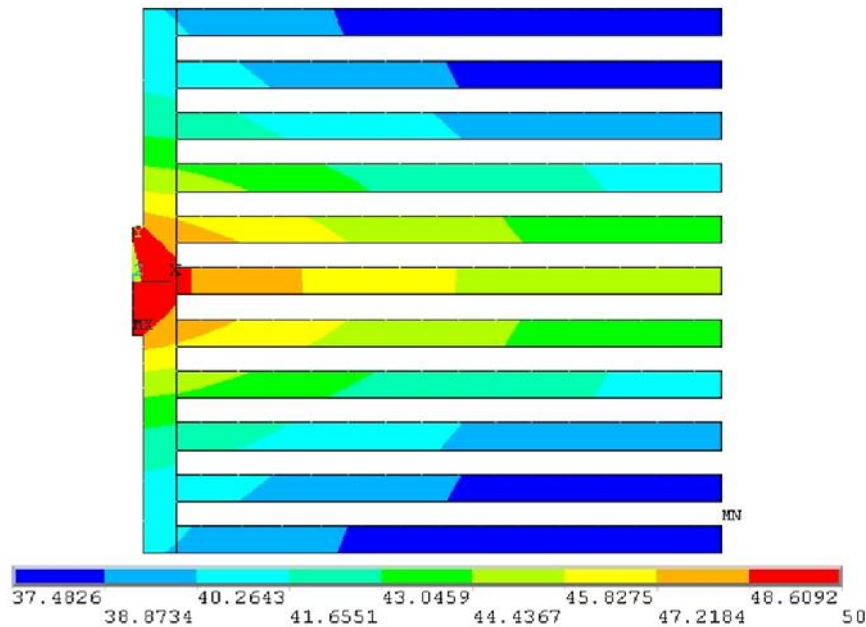


$$\frac{1}{R_{s \parallel}} = \frac{1}{R_1} + \frac{1}{R_2} + \frac{1}{R_3} + \frac{1}{R_4} + \dots + \frac{1}{R_{2n+1}}$$

Lienhard J., IV, Lienhard J., V., *A heat transfer textbook*. Cambridge, Massachusetts: Philogiston Press, 2012. Chapter 2.3

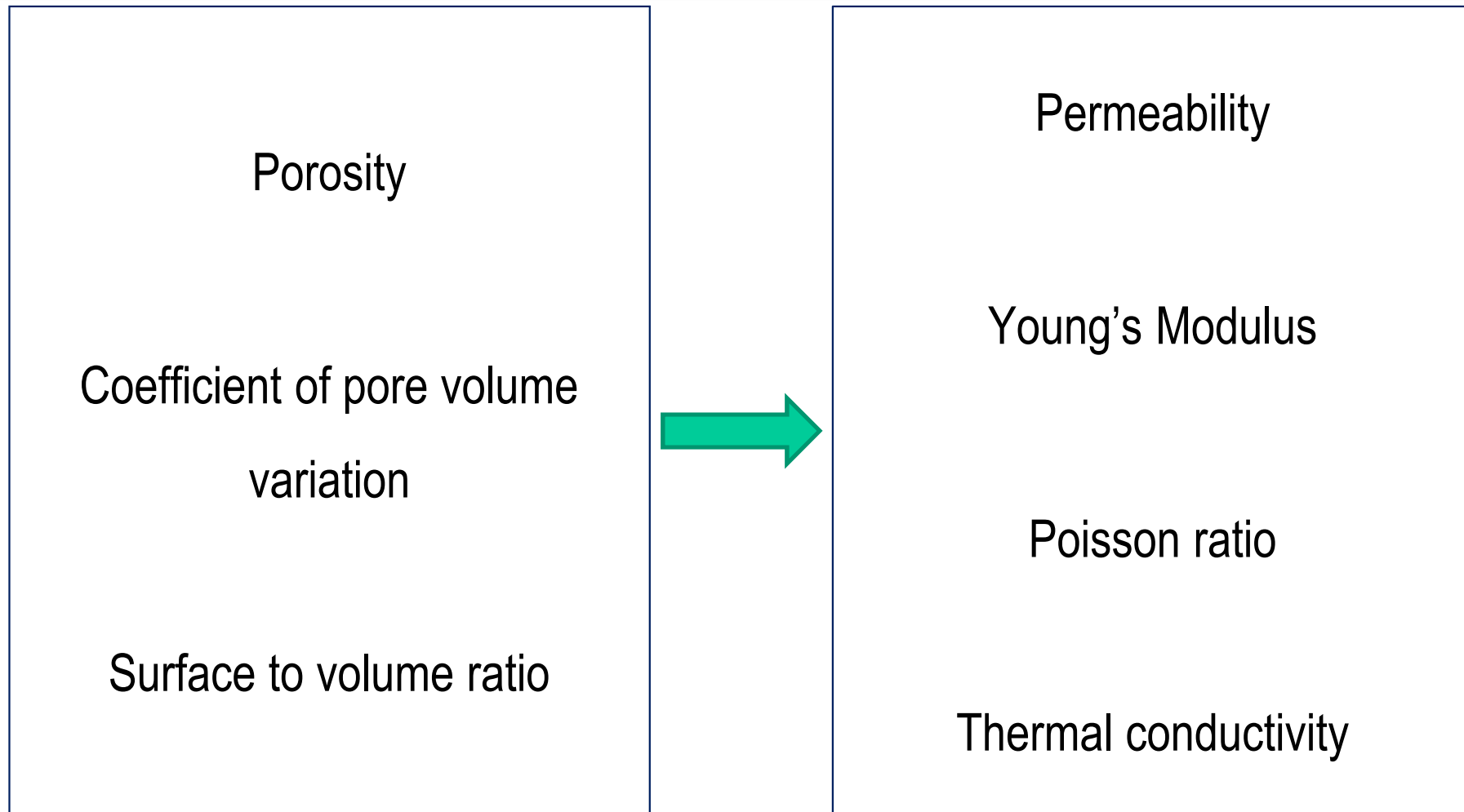


Heat sink



Cu thickness [A]	λ along layers [$\text{Wm}^{-1}\text{K}^{-1}$]	λ across layers [$\text{Wm}^{-1}\text{K}^{-1}$]	Heat rate [W/s]
1,00E+00	400	400	117,46
1,00E+06	400	392	117,44
1,00E+05	400,28	335,76	117,34
1,00E+04	402,76	137,3	116,92
9,90E+02	427	19,87	117,1
9,40E+01	676	2,08	121,76

Design of foam structures



Structural parameters of the designed structures

	Commercial alumina foam filter (10 pores per inch)	Designed structure with homogeneous pore volume distribution
Porosity [%]	75.01	78.17
Surface to volume ratio [1/mm]	0.77	0.76
Mean pore diameter [mm]	2.56	2.83

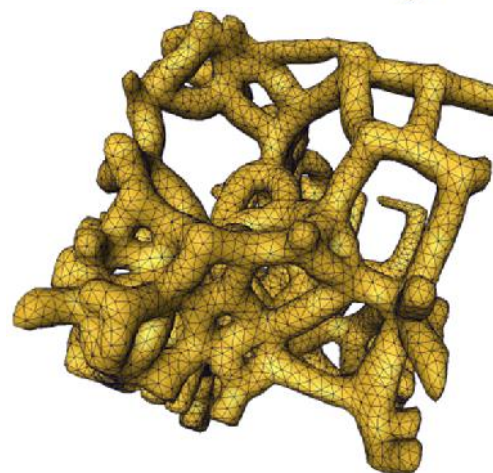
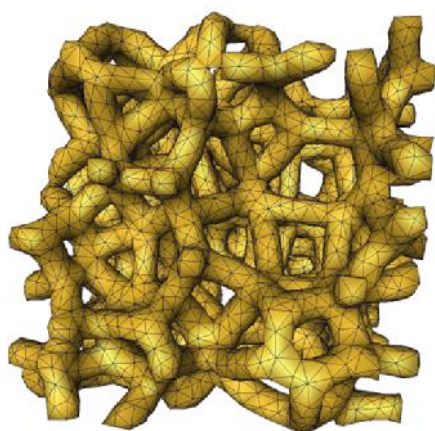
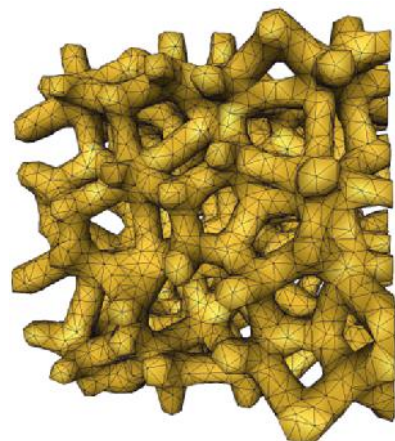
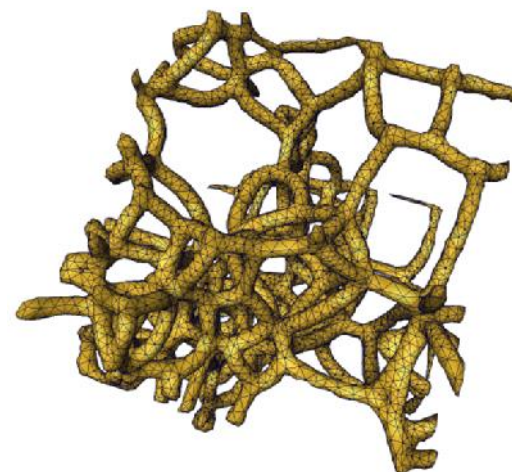
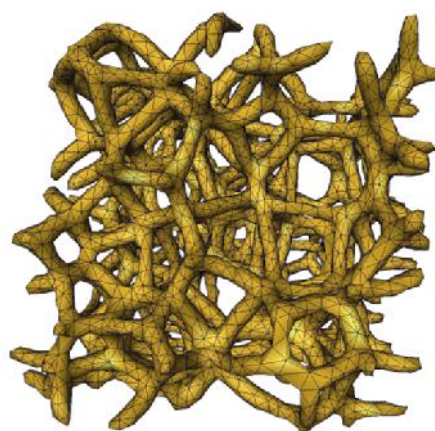
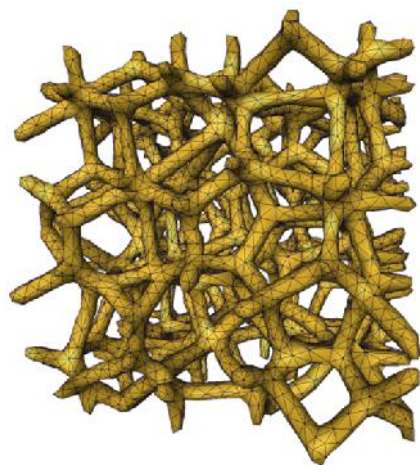
Examples of generated foam structures

$CV(V) = 0.45$

$CV(V) = 0.7$

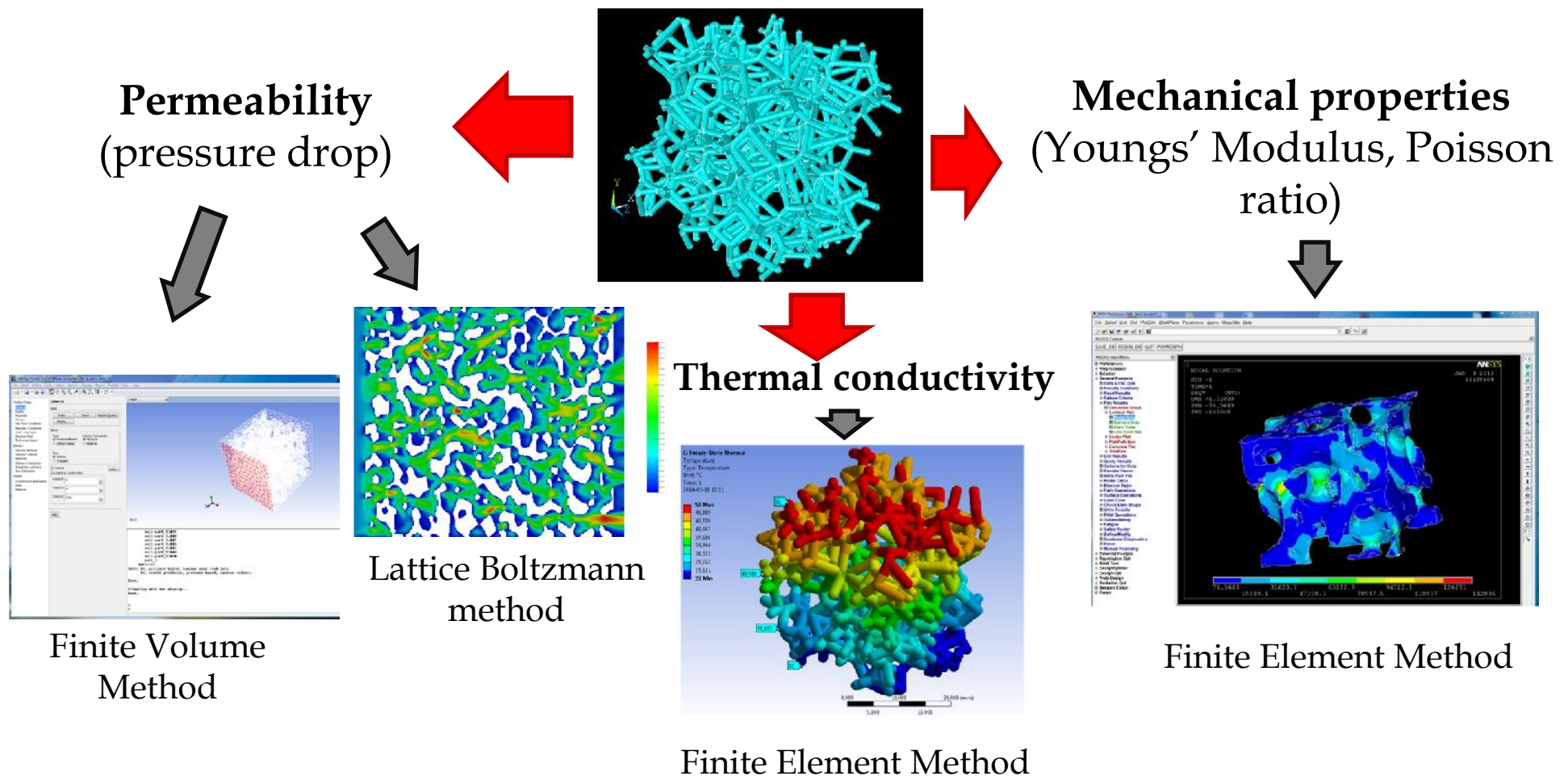
$CV(V) = 1.2$

Porosity ~ 90%

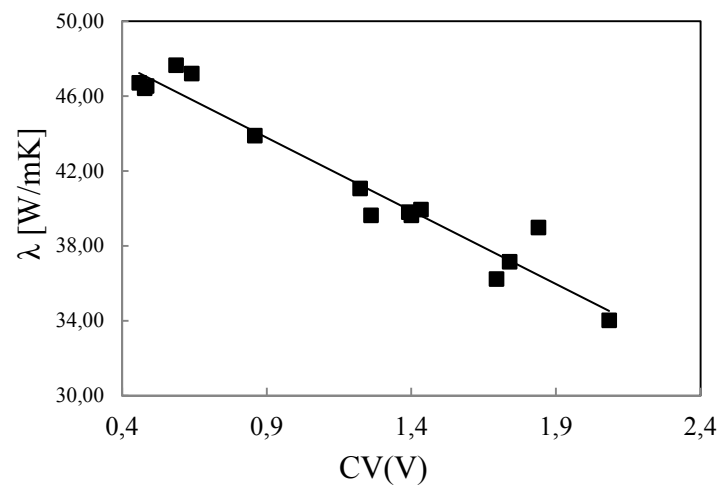
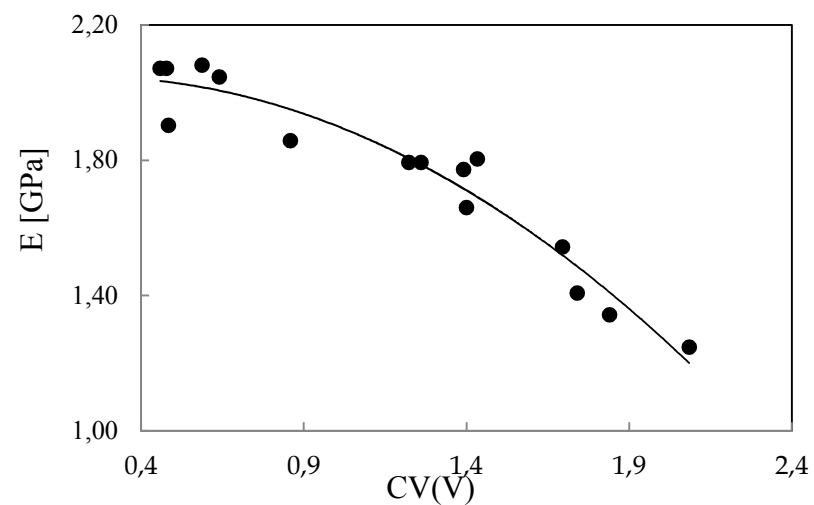
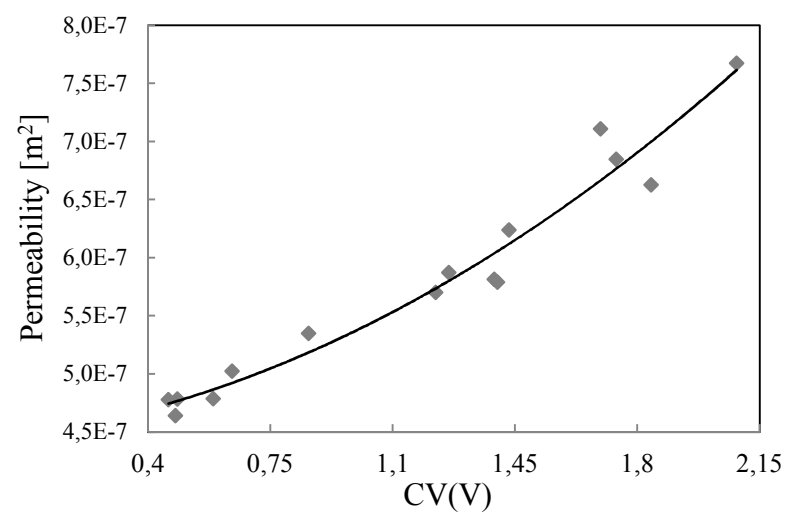


Properties of foam structures

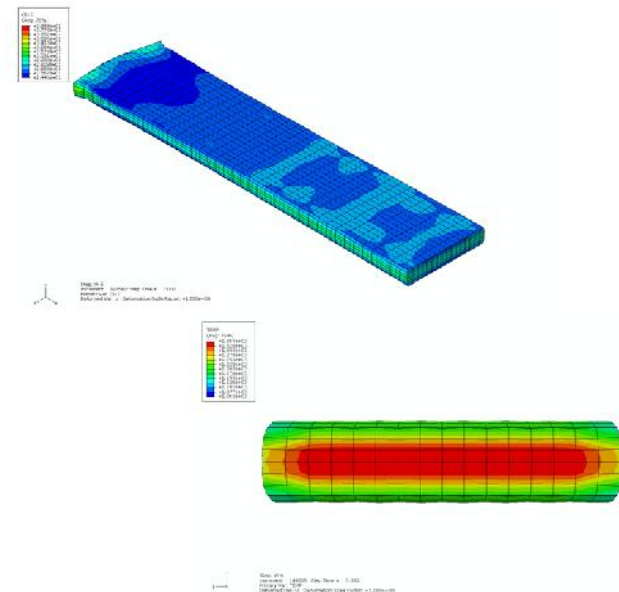
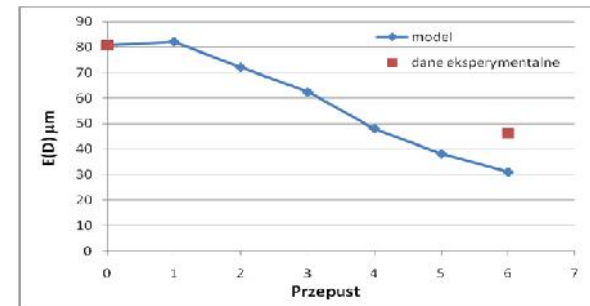
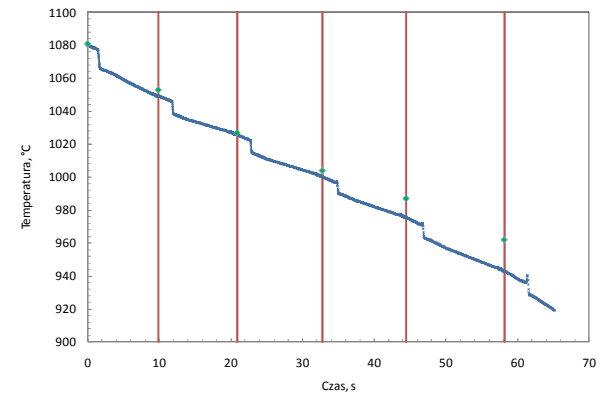
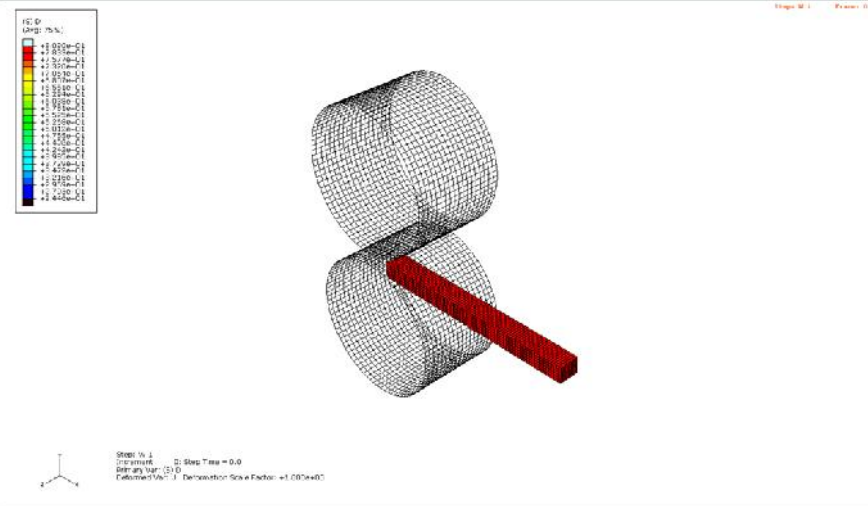
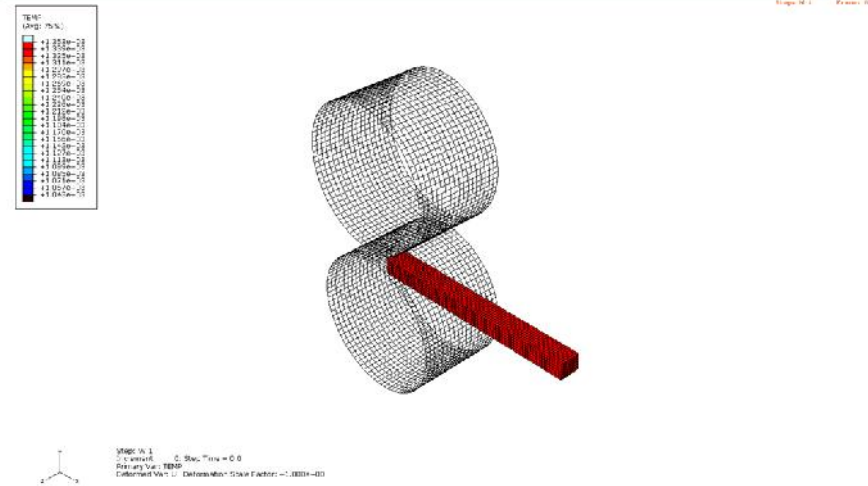
Analysis of material properties of the designed structures



Correlation of properties with structural parameters



Multi-pass hot-rolling of steel plates



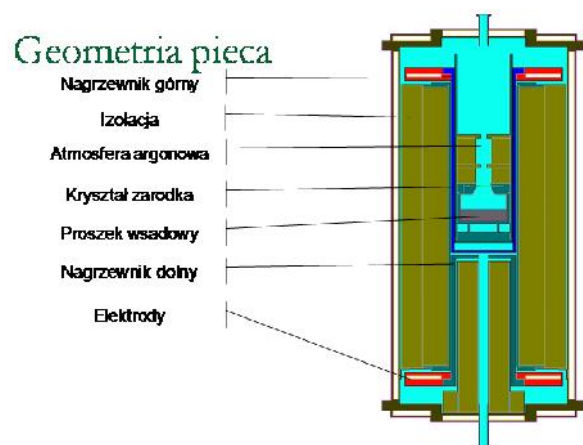
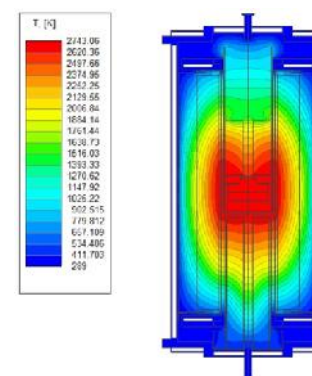


SiC PVT growth

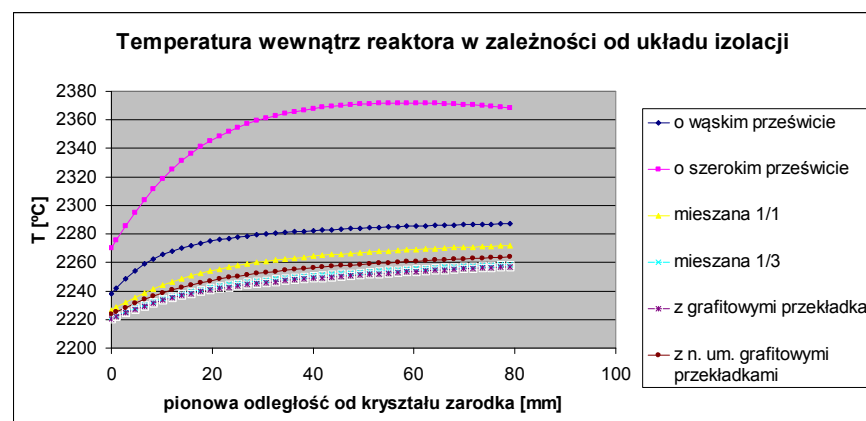


Heat and mass transport in PVT reactor

1. Reactor geometry
 - Thermal insulation
 - crystal thickness
2. Process conditions
 - temperature
 - pressure

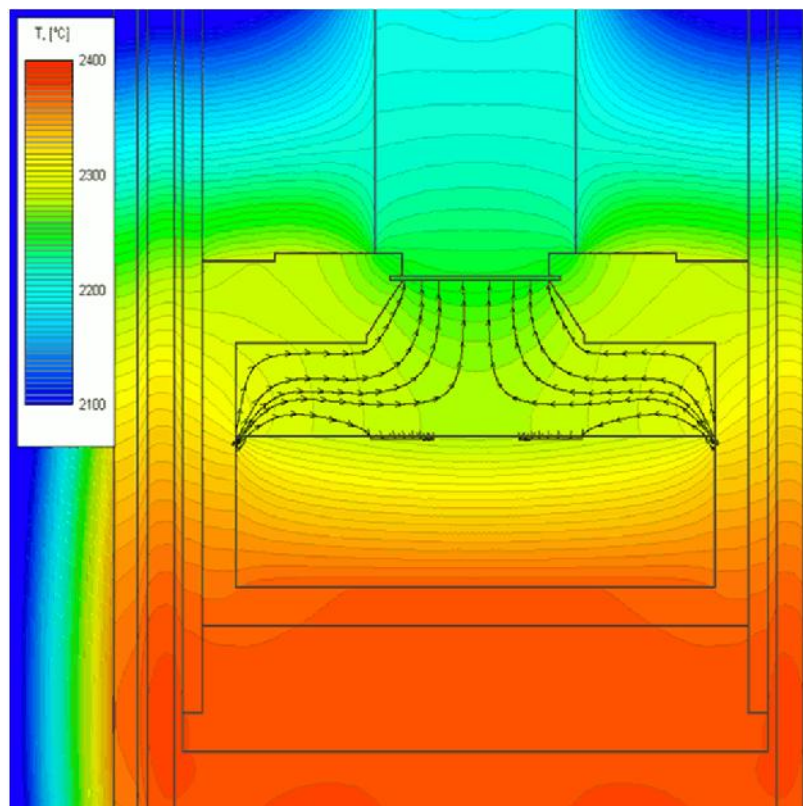


PVT reaktor - ITME

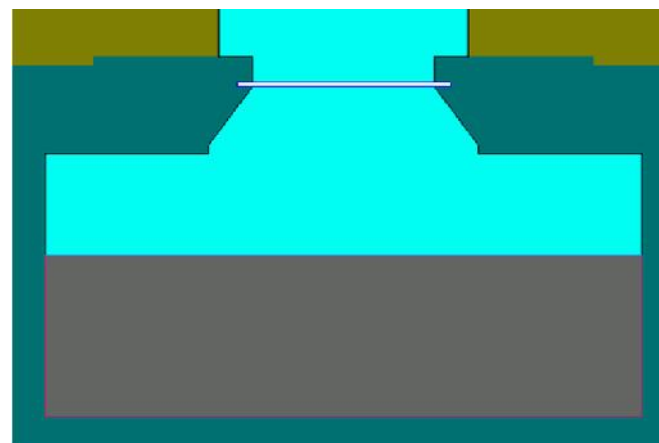




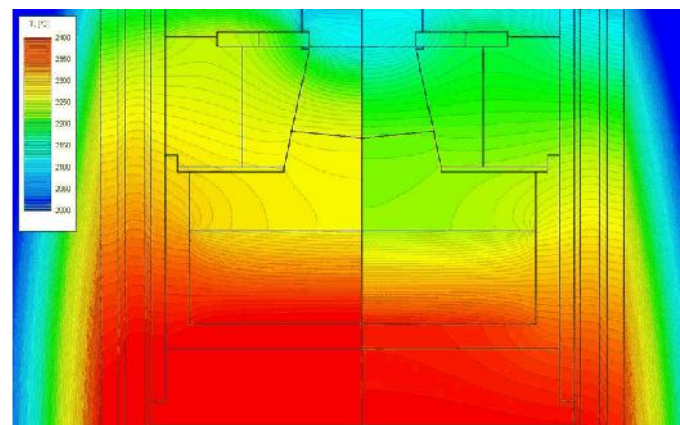
SiC PVT growth



Distribution of temperature



Evolution of dislocations



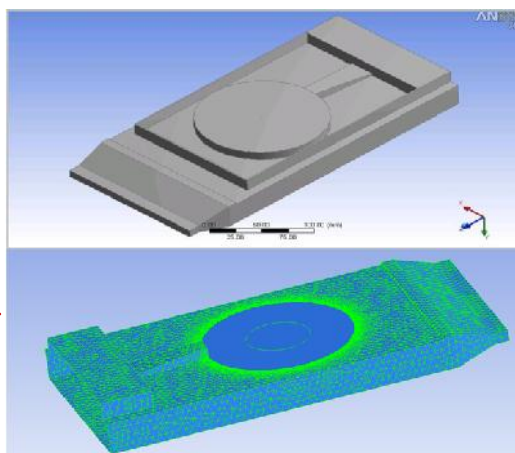
Crystal growth rate
(15d1 vs. 8d8)

GaN and Graphene epitaxial growth

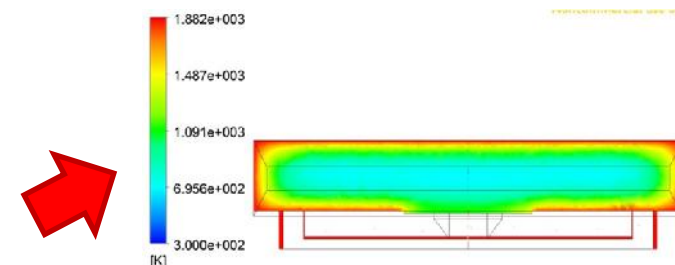
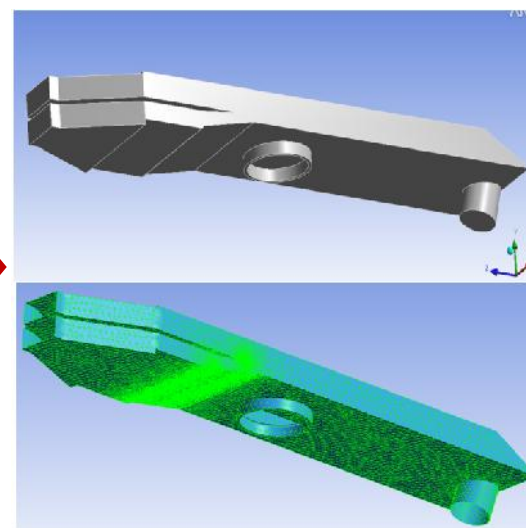


Graphene epitaxy

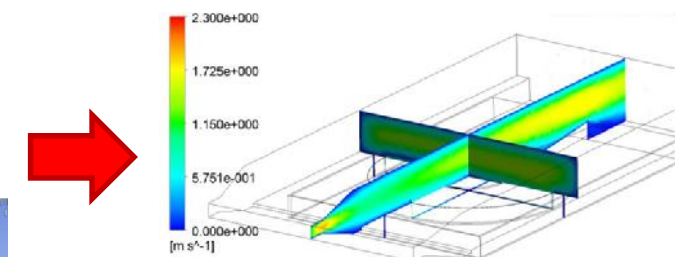
GaN epitaxy



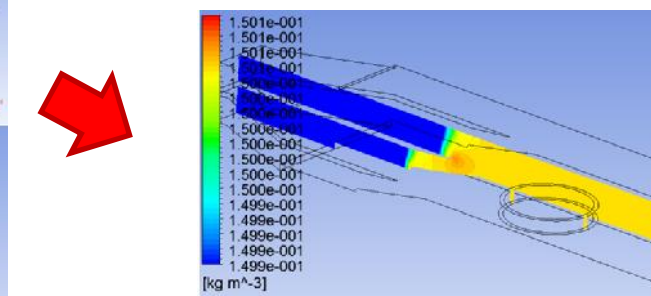
Numerical models



Temperature distribution

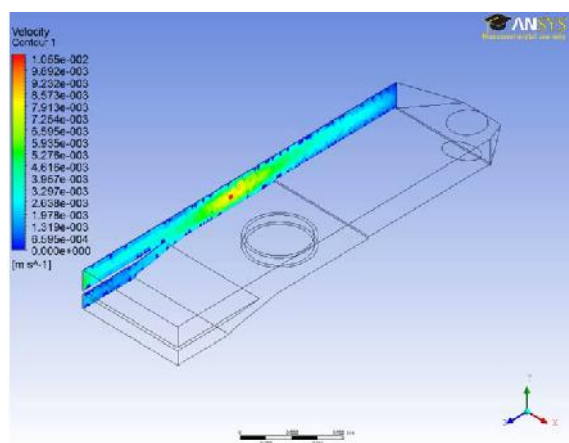


Gas velocity distribution

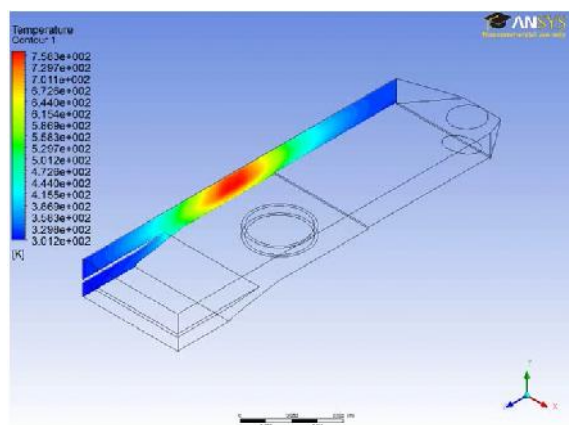


Density distribution

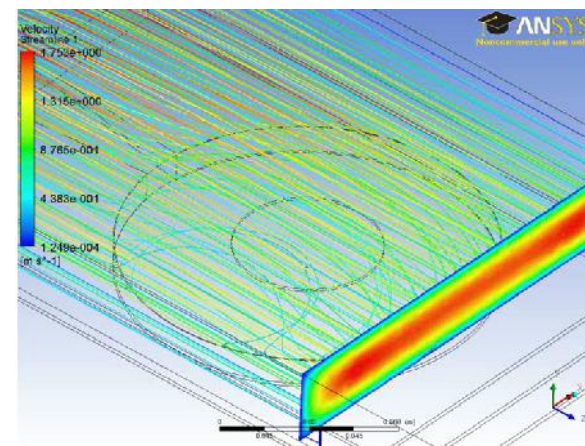
GaN and Graphene epitaxial growth



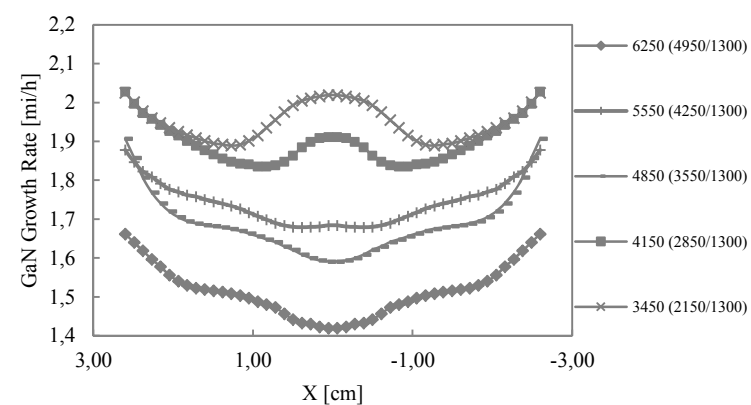
Velocity - MOVPE



Temperature - MOVPE



Velocity - CVD



GaN crystal growth rate

Summary

1. Many materials properties and processes can be nowadays modeled at various time and length scales
2. Software for atomic scale and continuum scale is well developed
3. There is still lack of software dedicated to meso-scale applications

Hardware and software



Local workstations

- 64-bit cluster (76 AMD-Opteron cores)
- 64-bit cluster (360 AMD-Opteron cores – 90 processors)
- 5 x workstations (2xAMD-Opteron Dual Core)

- Materials Studio (Accelrys)
- MedeA
- VASP
- Phonon
- Abinit
- CPMD
- IMD
- XMD
- LAMMPS
- Abaqus
- ANSYS
- Marc/Mentat
- Nastran/Patran
- Fluent/Fidap
- AMIRA
- Atomeye
- AVS